

# The role of length scales in material failure

Gergely Molnár

## Jury

Rapporteurs:

*Samuel Forest - Mines Paris PSL CNRS*

*David Rodney – UBCL1*

*Jean-François Molinari - EPFL*

Examinateuse:

*Laura De Lorenzis - ETH Zürich*

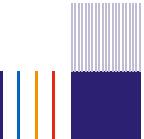
Examinateurs:

*Nicolas Moës - UCLouvain*

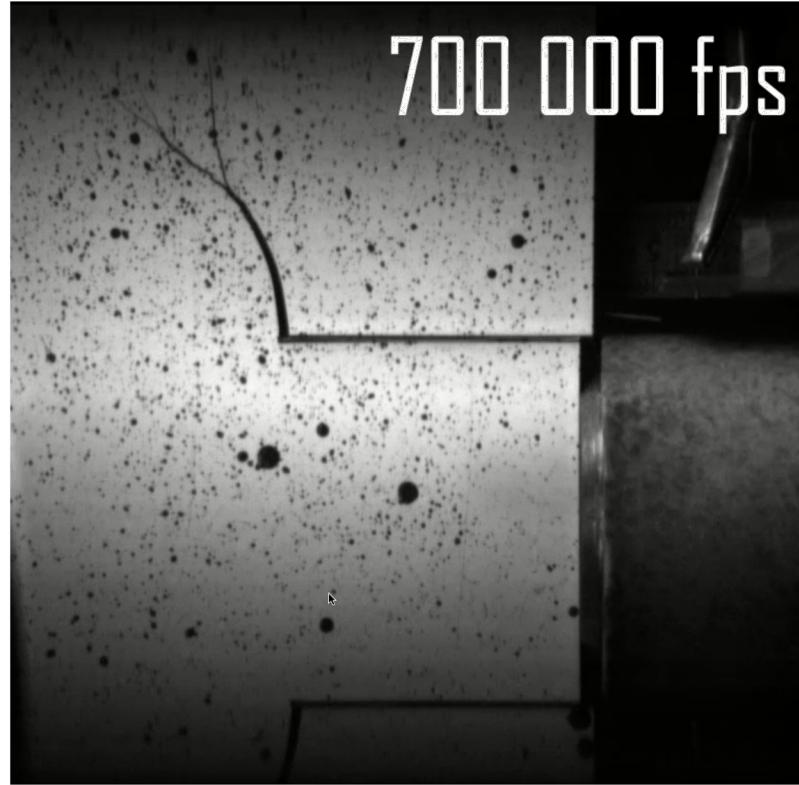
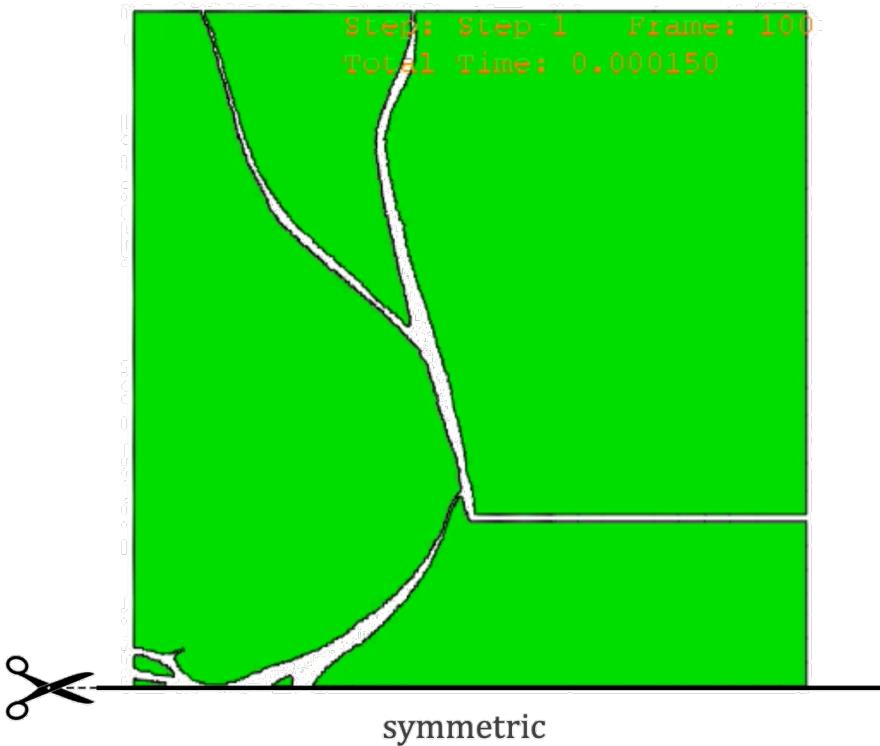
*Anthony Gravouil - INSA Lyon*

Lyon, France

05/03/2025



## Phase-field



Video and experiment by Rian Seghir (GeM)

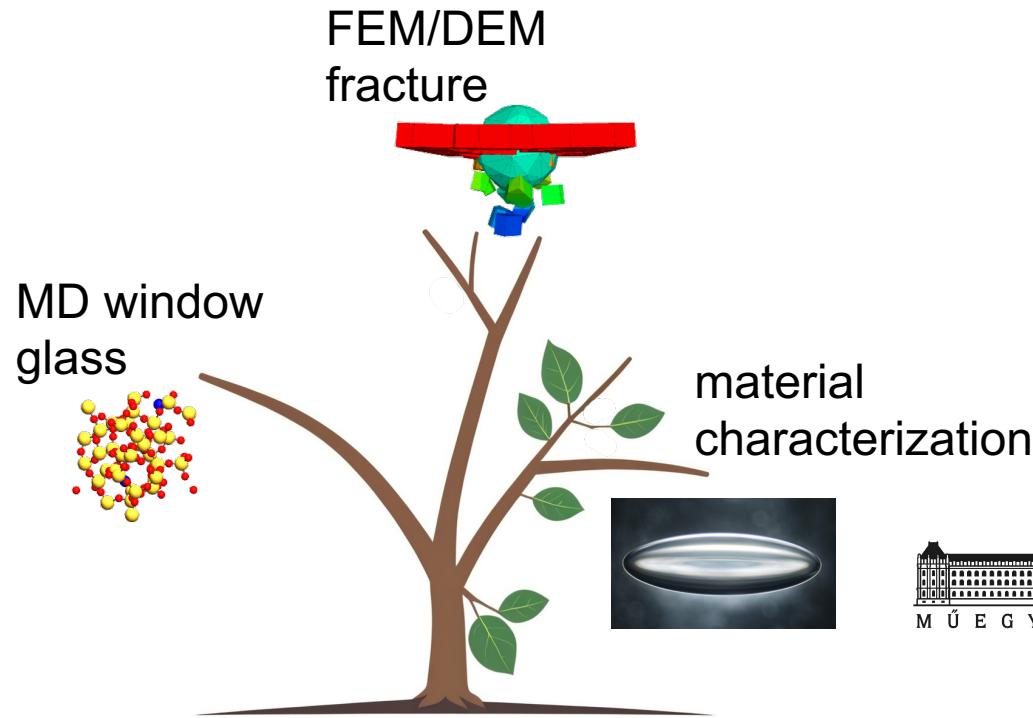
# Introduction

Ph.D.

## My journey

2011-2014

Multi-scale modelling of structural glass



MÁTRÉKA

MÁTRÉKA

# Introduction

Postdocs

2014-2018

## My journey

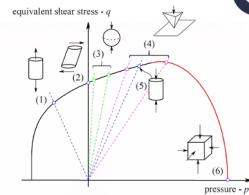
atomistic yield criterion



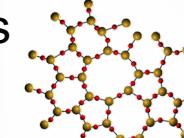
micro-scale analysis



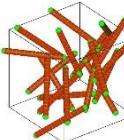
cellulose MD + DEM  
**UGA**



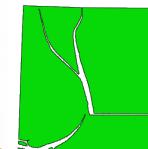
atomic scale simulations  
fracture



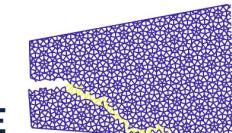
phase-field fracture



dynamic fracture

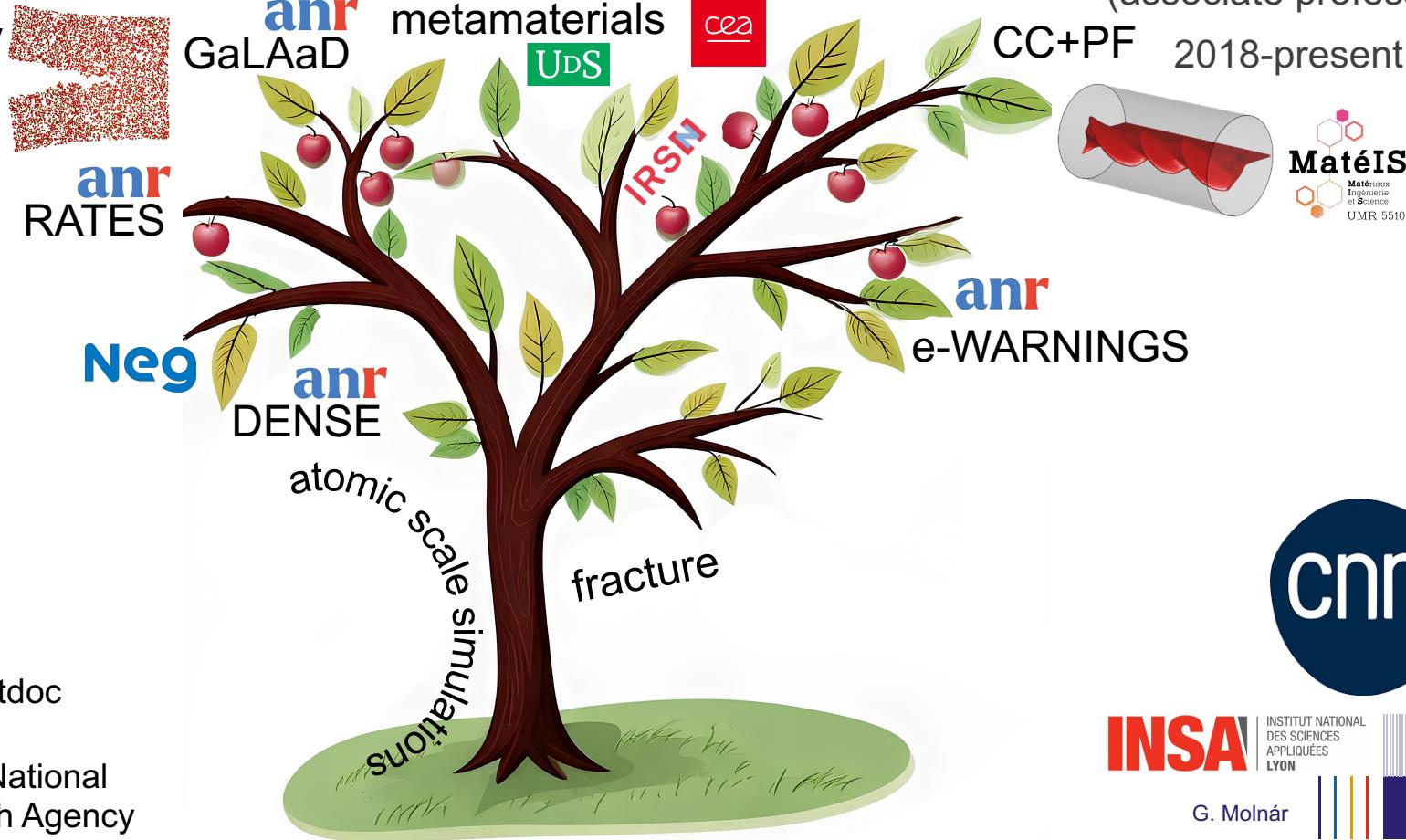


Cosserat fracture



# Introduction

My journey



CR CNRS

(associate professor)

CC+PF 2018-present

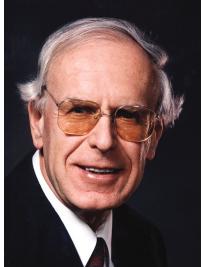
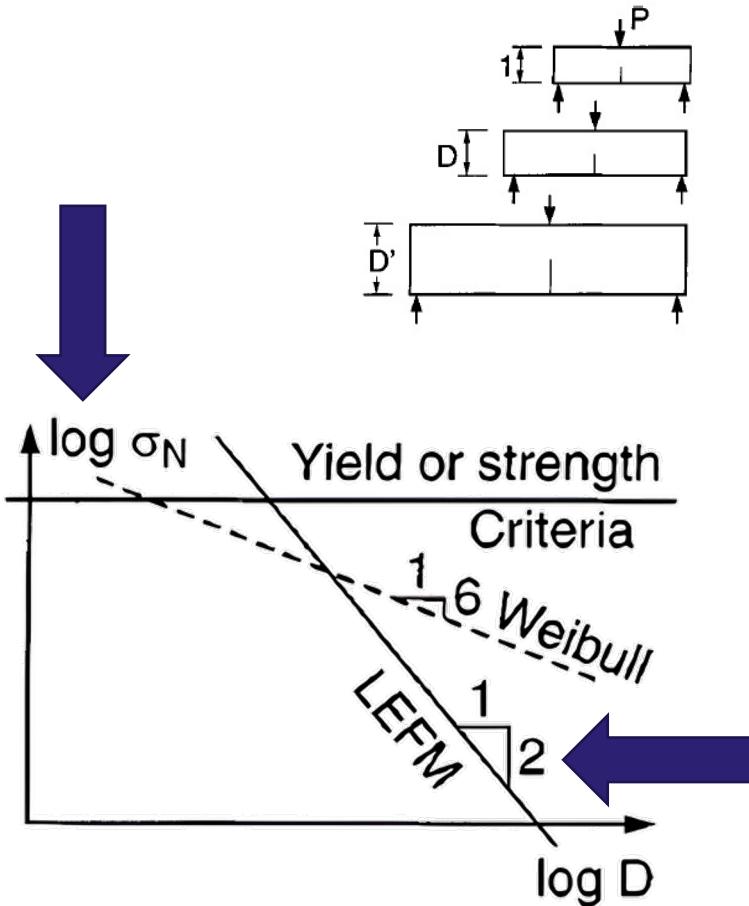


# Motivation

## Length scale in failure



Laboratory testing



Bažant et al. (1999)



Hoover dam

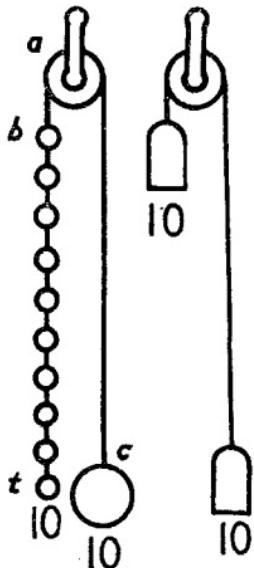
# Motivation

## History

1500's



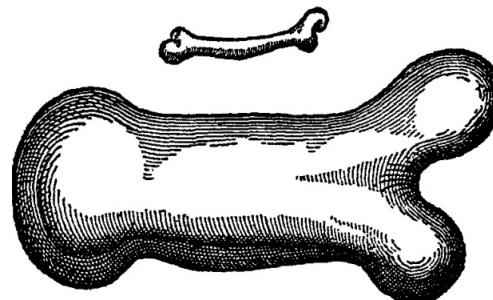
(Codex Atlanticus,  
Leonardo da Vinci)



"among cords of equal thickness the **longest** is the **least strong**..."

"a cord is so much **stronger**...as it is **shorter**"

(Two New Sciences,  
Galileo Galilei)

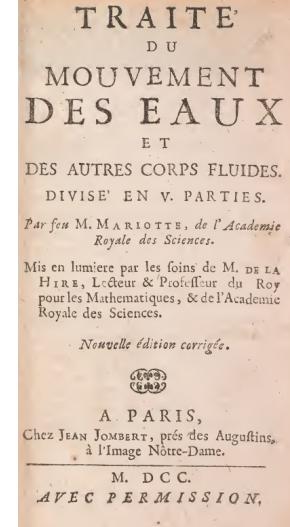


"...the **smaller** the body the  
**greater** its relative **strength**"

(Traité du mouvement  
des eaux, Edmé Mariotte)

1638

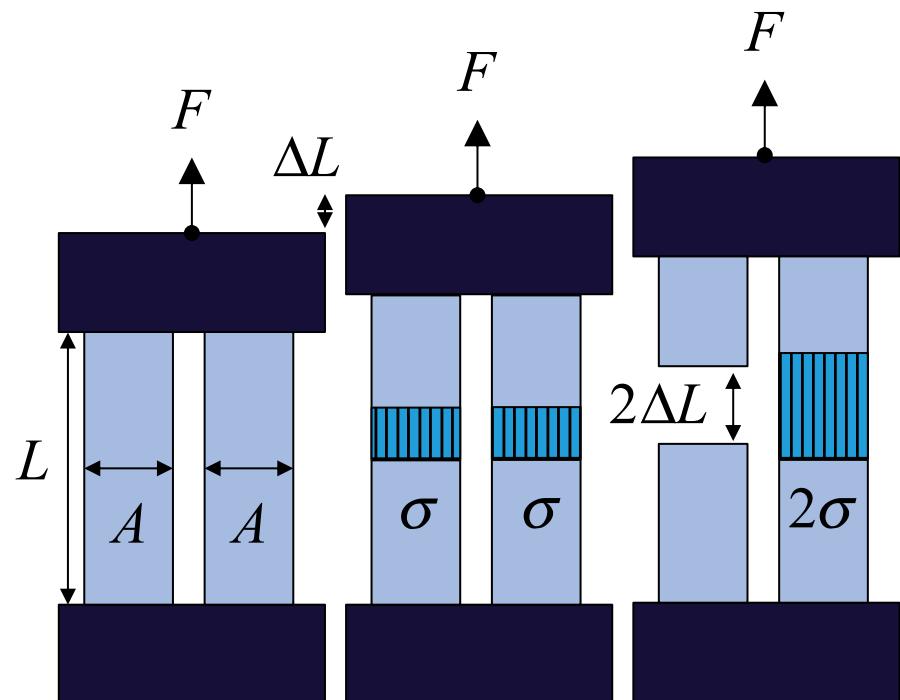
1686



"A long rope and a short one  
always support the same  
weight, unless that in a long  
rope there may happen to be  
some **faulty place**,  
in which it will break sooner  
than in a shorter."

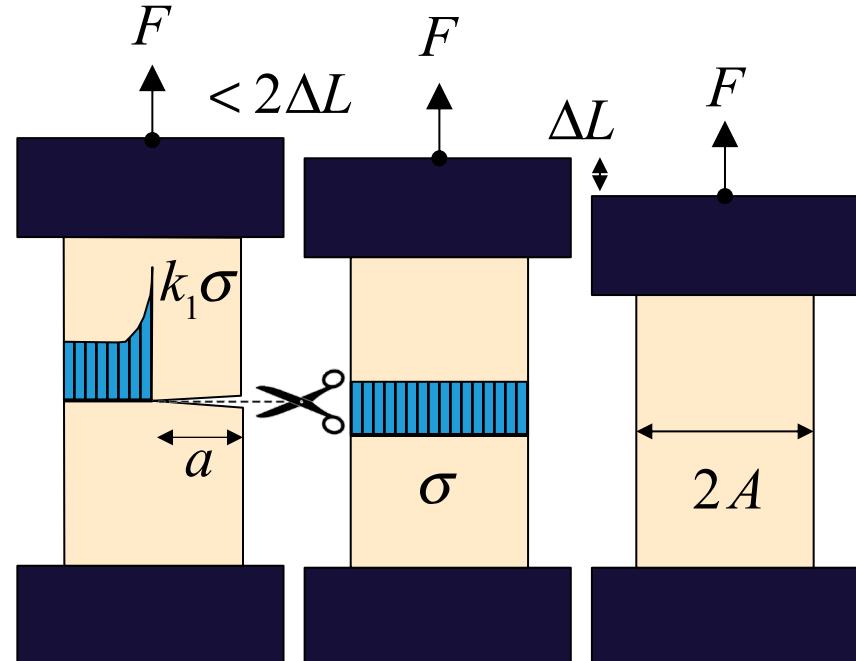
# Motivation

## Classical mechanics



$$\sigma = \frac{F}{2A}$$

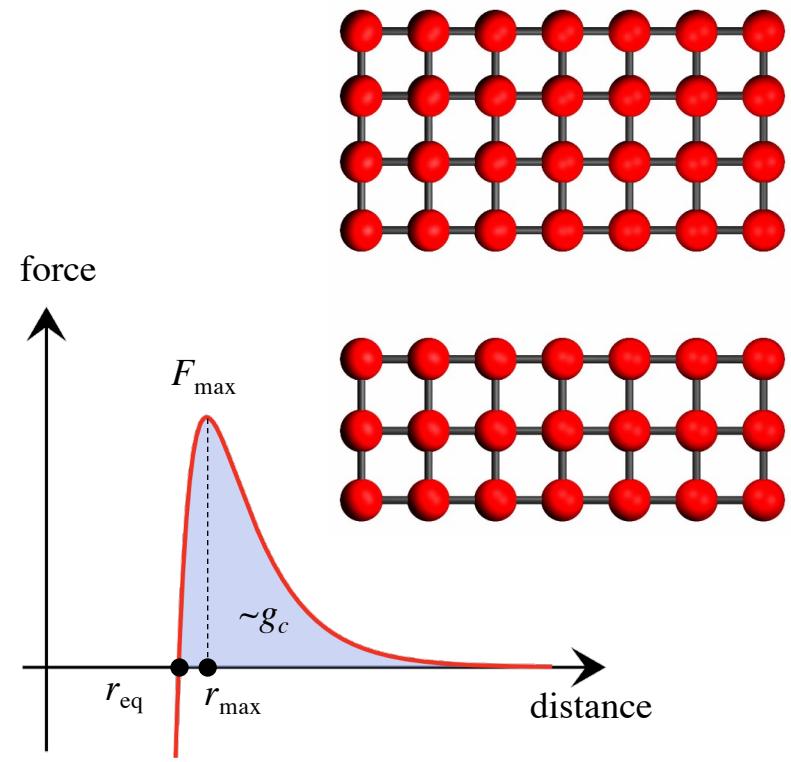
## Fracture mechanics



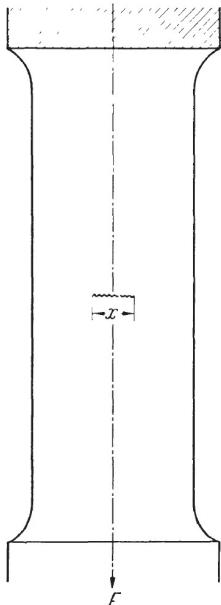
$$\sigma = \frac{F\sqrt{\pi a}}{2A} \frac{1}{\sqrt{2\pi r}} g(\theta)$$

# Motivation

## Energy release rate / Toughness

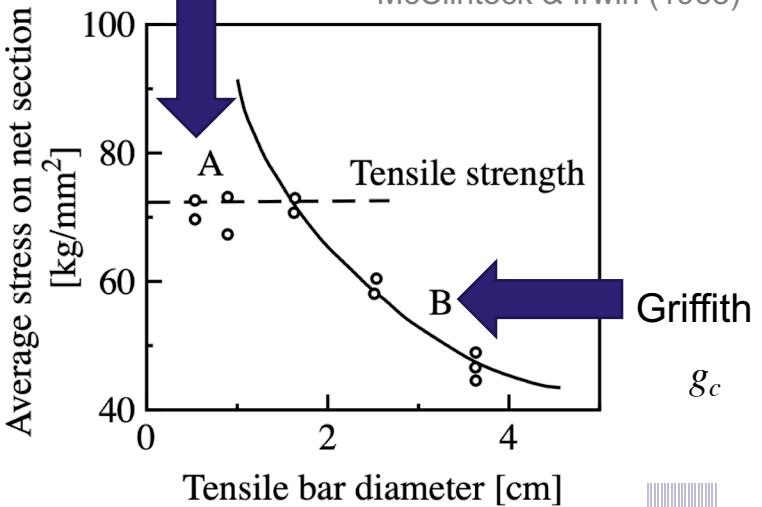


## Limitations



Irwin's length

strength



$$l_{mat} = \frac{Eg_c}{\sigma_c^2}$$

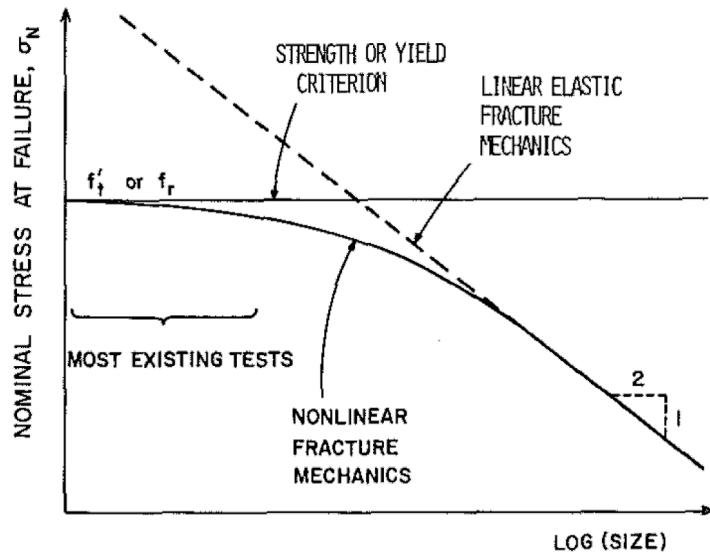
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# Motivation

## Length scale in failure

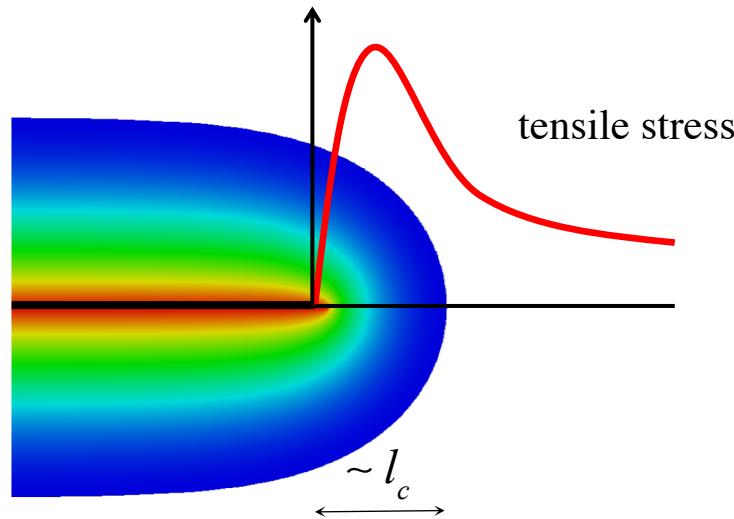


(Bažant et al., 1984)

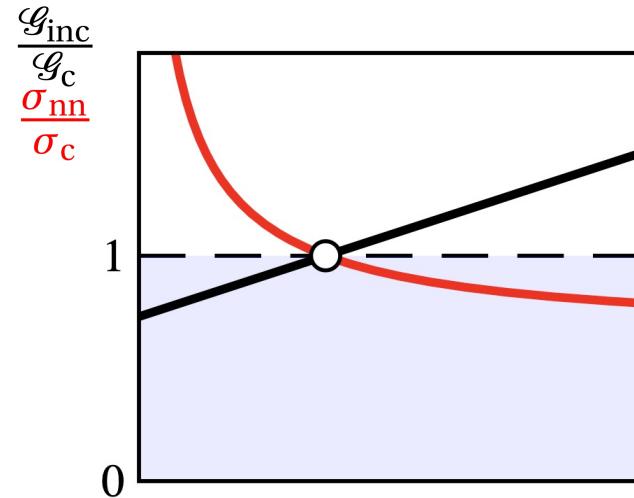
- concrete Shah & Swartz (1987)
- composites Bažant et al. (1996)
- rocks Bažant et al. (2004)
- ceramics Usami et al. (1986)
- silica glass Luo et al. (2016)
- wood Simon (2009)
- SiC Bažant et al. (1990)
- sea ice Dempsey et al. (1995)
- alumina Leguillon et al. (2018)
- etc...

# Continuum Fracture

## Phase-field model

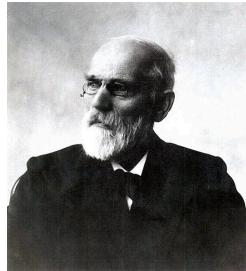


## Coupled criterion



# Continuum Fracture

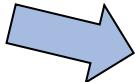
## Phase-field approach



liquid–gas interfaces

with **density functions**

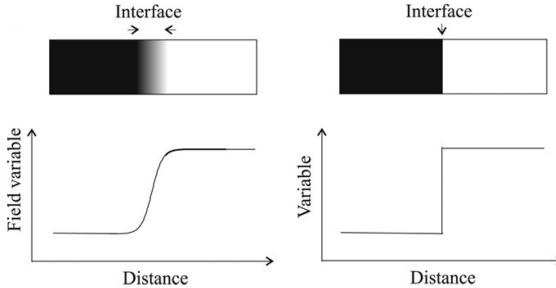
van der Waals (~1900)



**superconductivity**  
(Ginzburg & Landau, 1950)

**diffuse interfaces**

(Cahn & Hilliard, 1958)



L. M. Kachanov (1958)

continuum damage mechanics

$$\psi = \frac{1}{2} (1 - d) \epsilon_{ij} C_{ijkl} \epsilon_{kl}$$



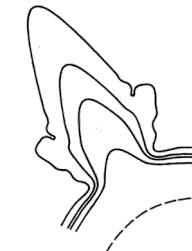
- fracture mechanics

(Bourdin et al., 2000)

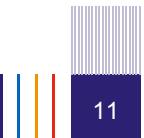
- solidification

- image processing

(Fix, 1983)  
(Langer, 1986)



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# Continuum Fracture



## Phase-field approach

1. Brittle fracture

$$-\frac{\partial \psi}{\partial a} = \frac{\partial S}{\partial a} = \mathbf{g}_c \quad (\text{Griffith, 1920})$$

(Bourdin et al., 2000)

2. Minimization problem

$$\Pi(\mathbf{u}, \Gamma) = \int_{\Omega} \psi(\varepsilon(\mathbf{u})) d\Omega + g_c \int_{\Gamma} d\Gamma$$

(Mumford & Shah, 1989)  
(Francfort & Marigo, 1998)

3. Crack energy density

$$\Pi(\mathbf{u}, \mathbf{d}) = \int_{\Omega} g(\mathbf{d}) \psi_0(\varepsilon(\mathbf{u})) d\Omega + \int_{\Omega} \frac{g_c}{c_{\omega} l_c^2} \left( \omega(\mathbf{d}) + l_c^2 |\nabla \mathbf{d}|^2 \right) d\Omega$$

crack energy  
density

$l_c \rightarrow 0$     $\Gamma$  converges

(Ambrosio & Tortorelli, 1990)  
(Bourdin et al., 2000)  
(Amor et al., 2009)  
(Miehe et al., 2010)

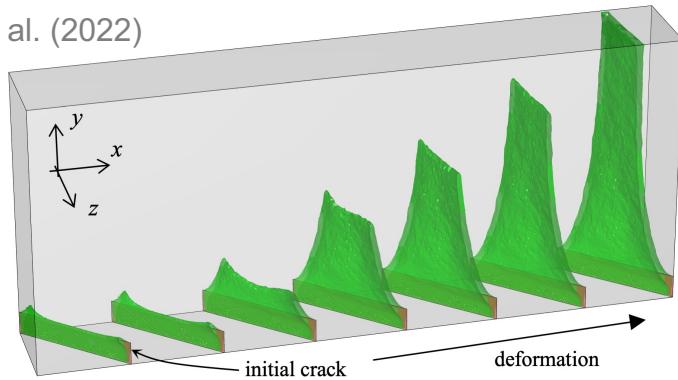


# Continuum Fracture

Molnár et al. (2017)

## Phase-field approach - Examples

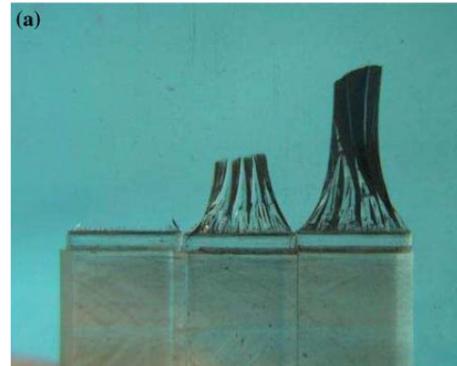
Molnár et al. (2022)



Solving **fracture mechanics** problems with Partial Differential Equations (**PDEs**)

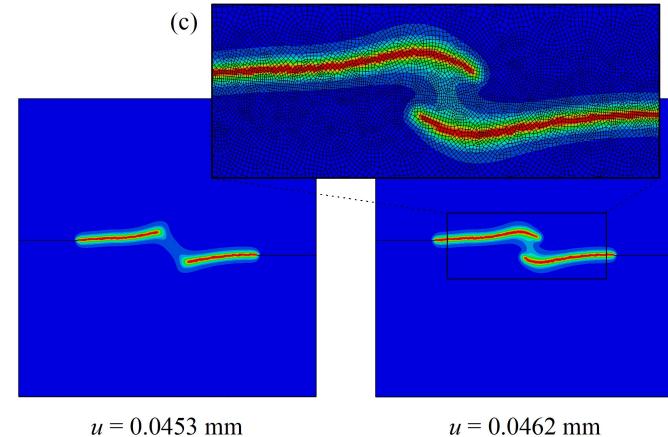
**Mode I+III**

Lazarus et al. (2008)



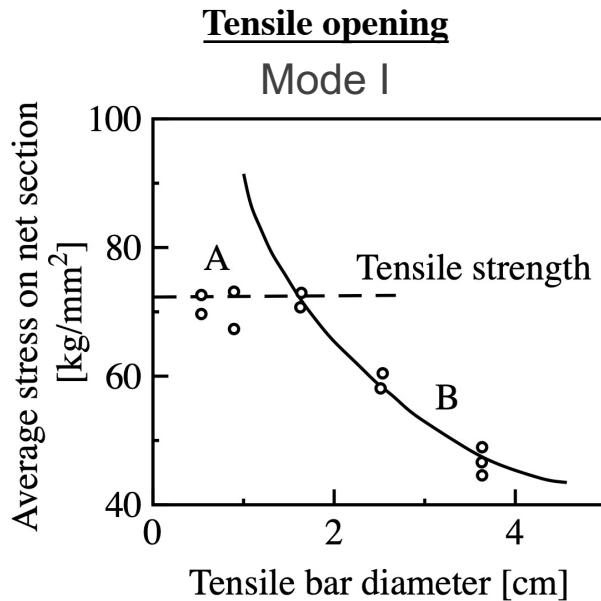
**Mode I+II**

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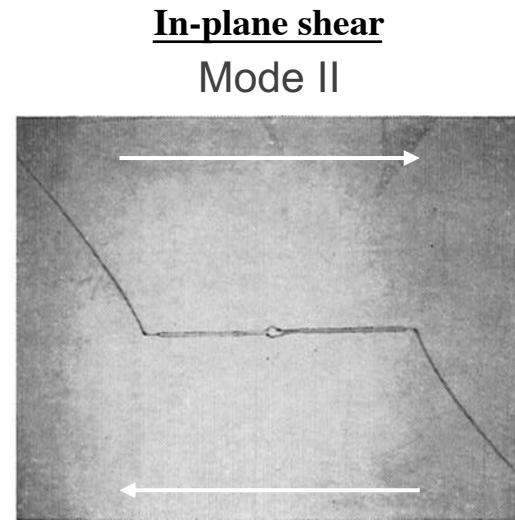


# Continuum Fracture

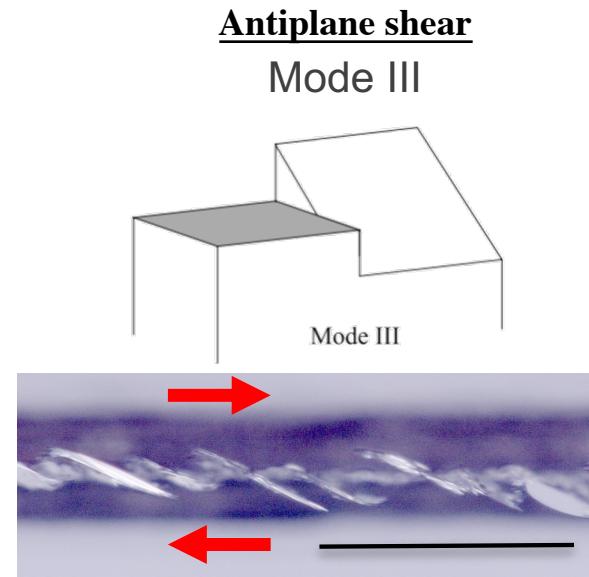
## Length scale in fracture



(Irwin, 1958)  
(Lubahn, 1956)



(Erdogan & Sih, 1963)



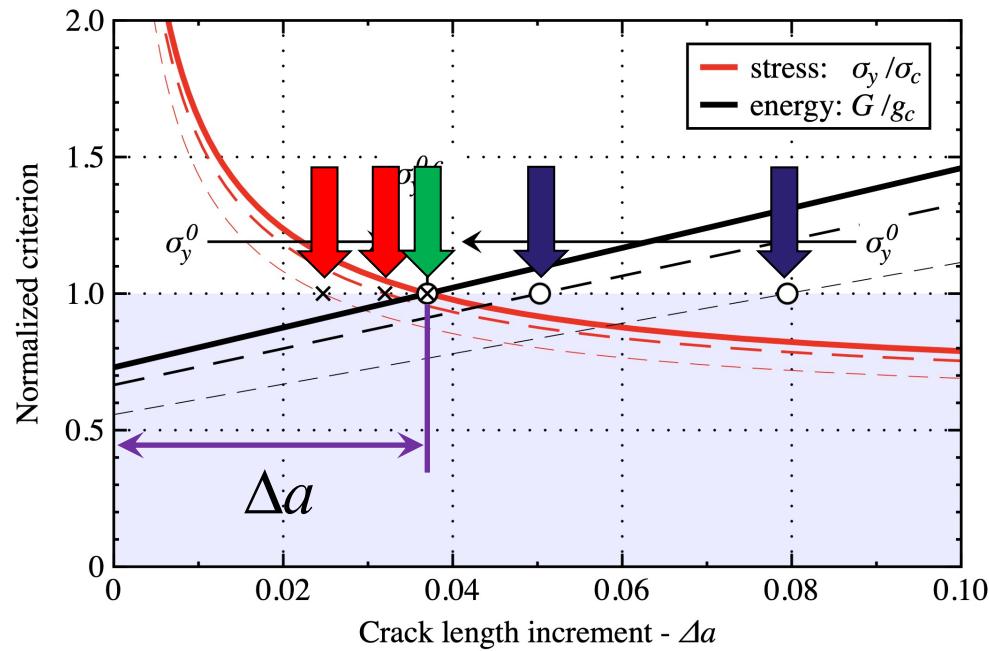
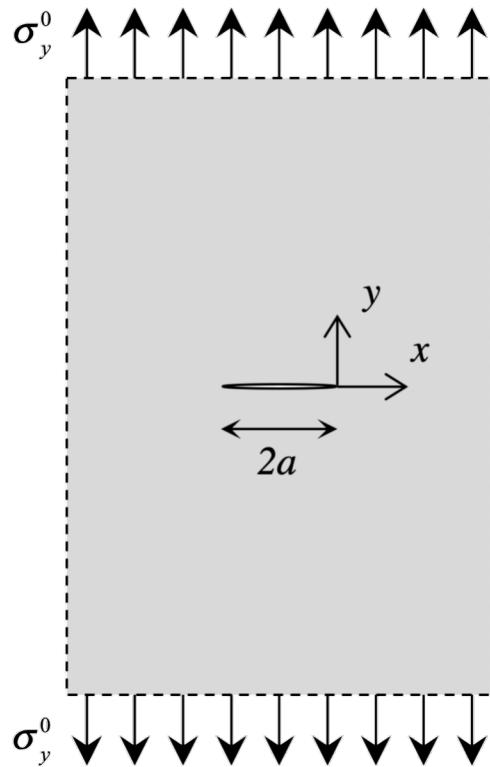
(Sommer, 1969)  
(Chen et al., 2015)

# Continuum Fracture



Leguillon (2002)

## Finite fracture mechanics



$$\frac{\partial \Pi}{\partial a} \geq g_c$$

Griffith

Stress

$$\sigma_1(a + \Delta a) \geq \sigma_c$$

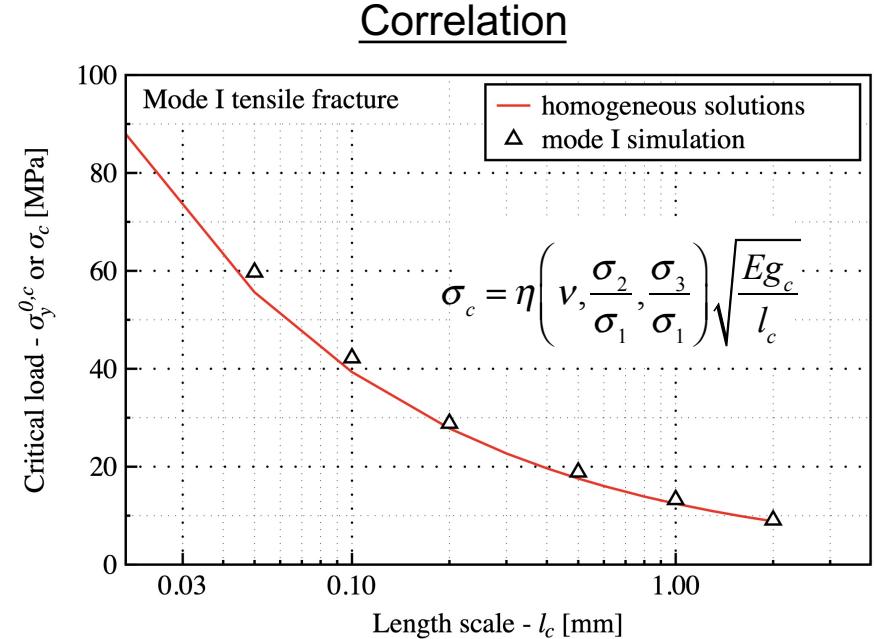
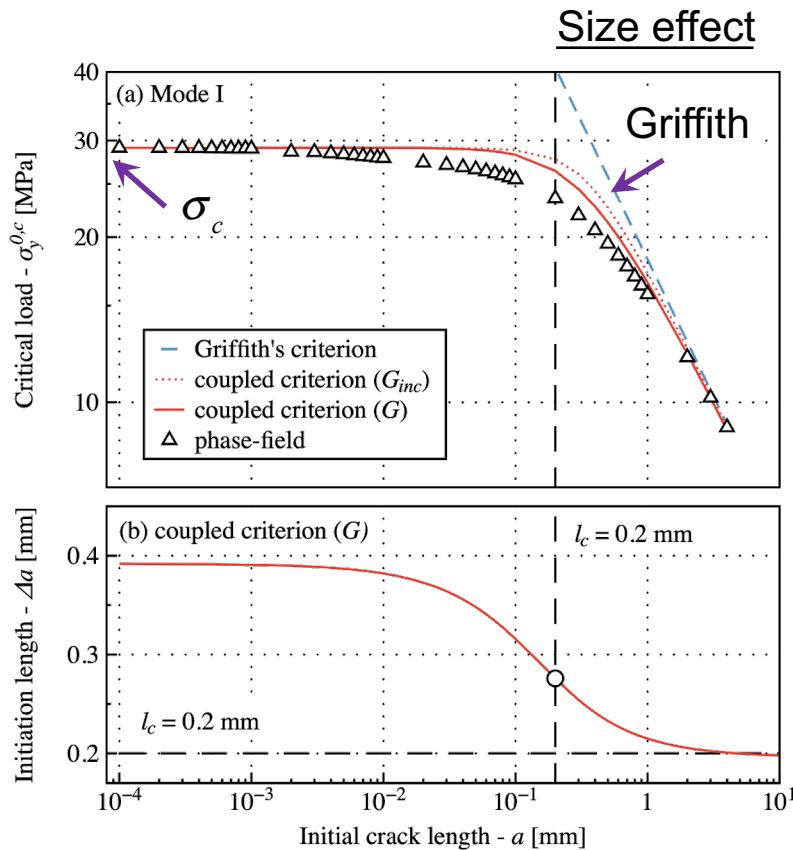
Molnár et al. (2020)

Coupled **stress** and **energy** criteria

# Continuum Fracture

Molnár et al. (2020)

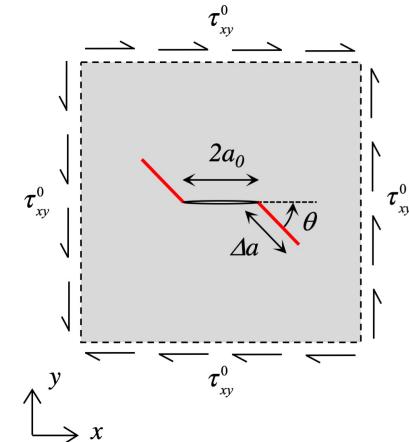
## Tensile opening



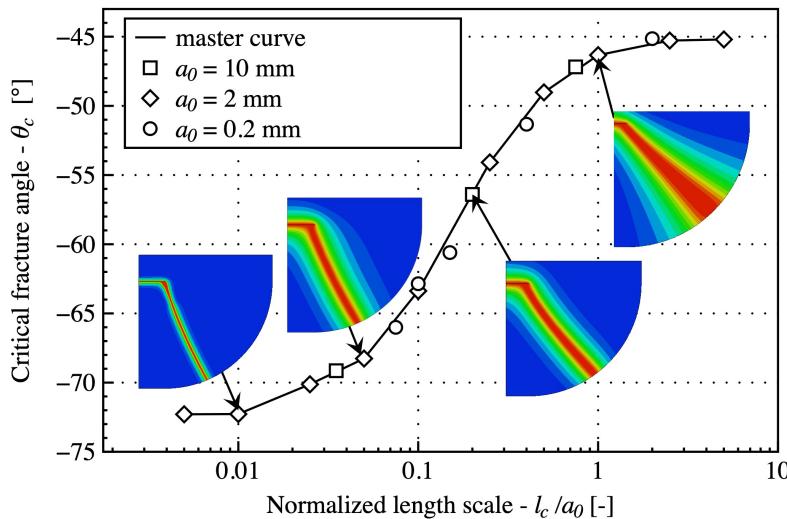
# Continuum Fracture

## In-plane shear fracture

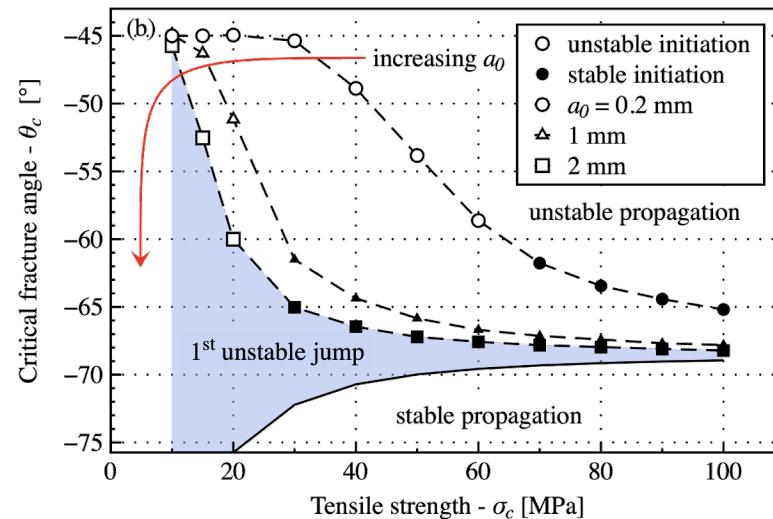
Molnár et al. (2020)



Phase-field



Coupled criterion



# Continuum Fracture

## Antiplane echelon cracking

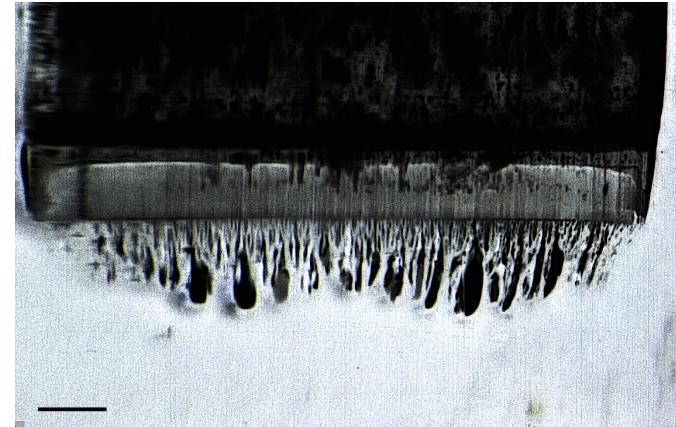
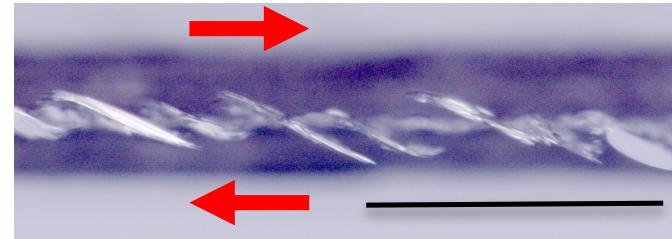
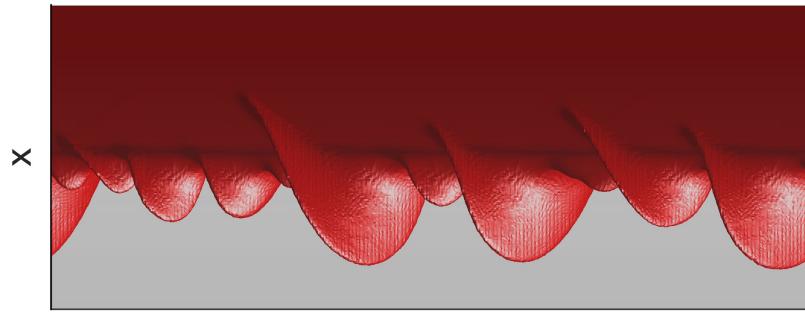
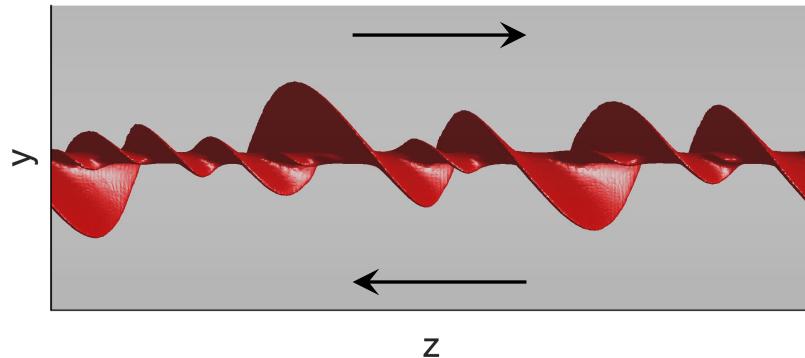


*Madeira, Portugal (2022)*



# Continuum Fracture

## Antiplane echelon cracking

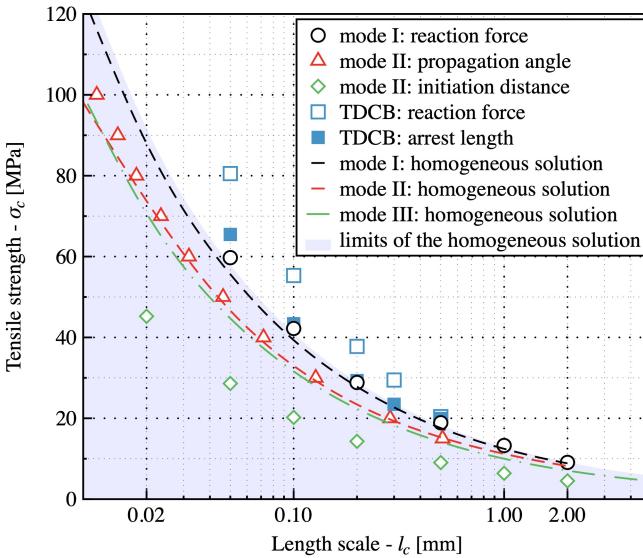


# Continuum Fracture

Molnár et al. (2020)  
Molnár et al. (2024)

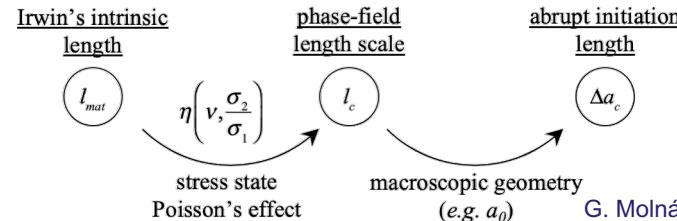
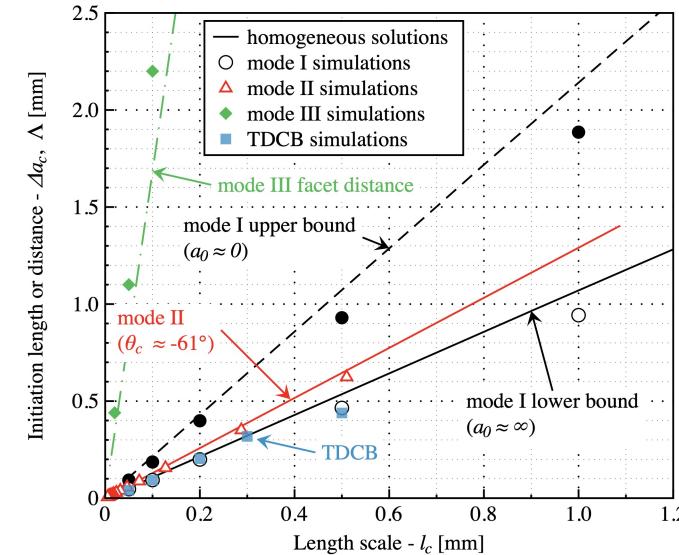
## Length vs strength

### Strength



$$\sigma^{\max} = \eta \left( v, \frac{\sigma_2}{\sigma_1}, \frac{\sigma_3}{\sigma_1} \right) \sqrt{\frac{E g_c}{l_c}}$$

### Initiation length



# Continuum Fracture

(PMMA,  $v = 25 \text{ m/s}$ )

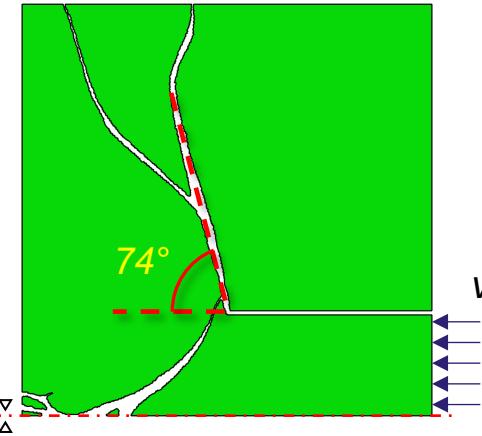
## Dynamic fracture

Kinetic energy

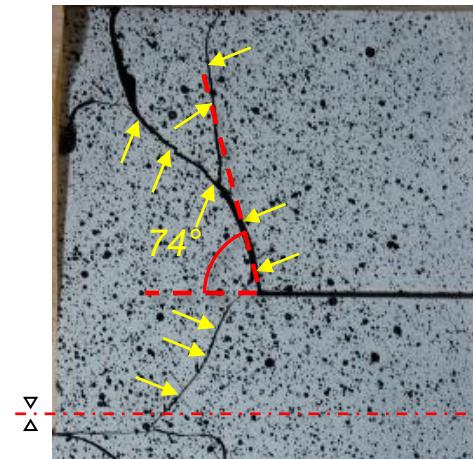
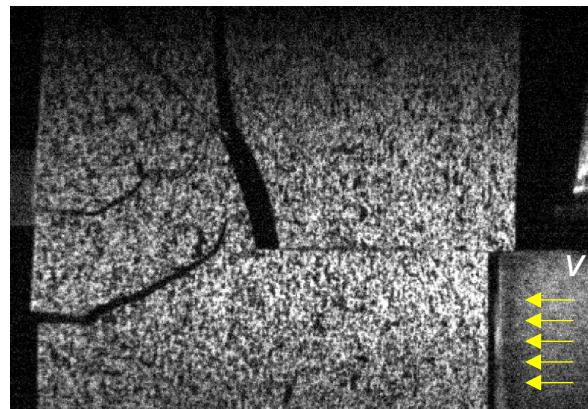
$$\mathcal{L} = D(\dot{\mathbf{u}}) - \Pi(\mathbf{u}, \mathbf{d})$$

$$D(\dot{\mathbf{u}}) = \frac{1}{2} \int_{\Omega} \dot{\mathbf{u}}^T \dot{\mathbf{u}} \rho d\Omega$$

(Molnár et al., 2022)



material  
 $E = 6 \text{ GPa}$   
 $g_c = 600 \text{ J/m}^2$

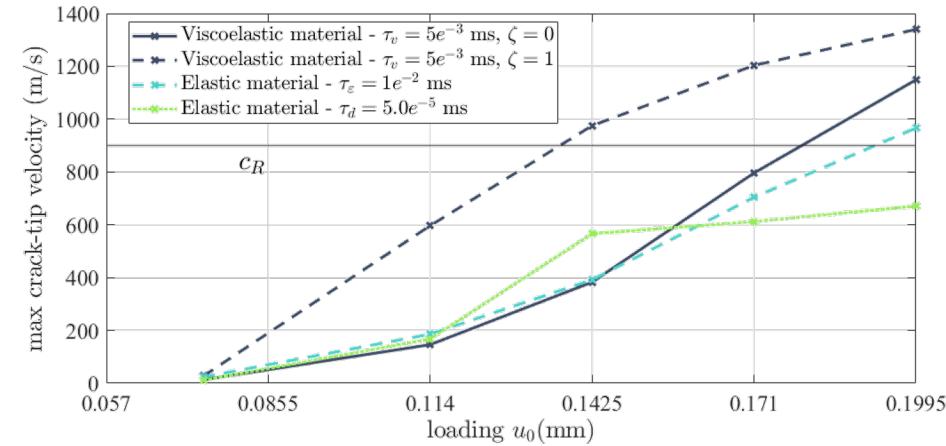


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# Continuum Fracture

## Dynamic fracture - viscoelasticity

### Crack tip velocity

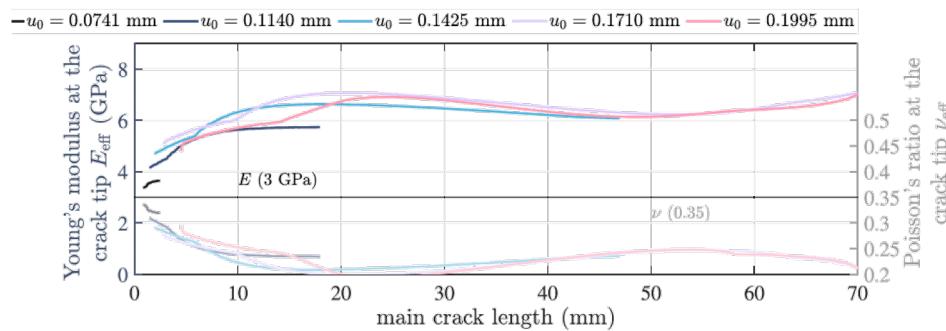
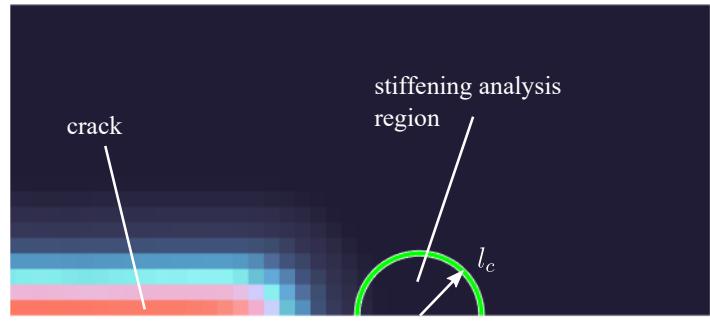


$$\Pi(\mathbf{u}, \mathbf{d}) = \Pi_{el}(\mathbf{u}, \mathbf{d}) + \Pi_d(\mathbf{d}) +$$

$$+ \int_{-\infty}^t \int_{\Omega} \boldsymbol{\sigma}_v(\dot{\boldsymbol{\epsilon}}_v, \mathbf{d}) : \dot{\boldsymbol{\epsilon}}_v d\tau d\Omega$$

Eid et al. (2023)

### Local stiffness

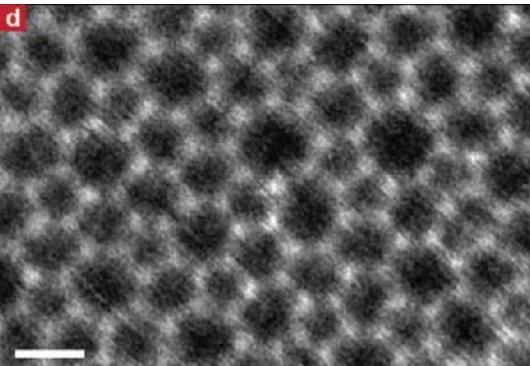
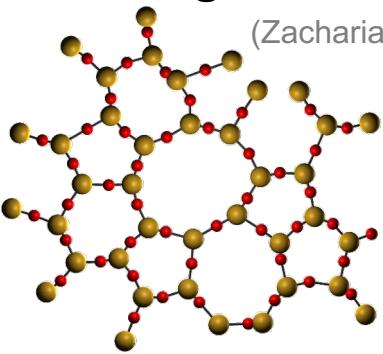


# Resistance of Silicate Glasses

Molnár et al. (2016)

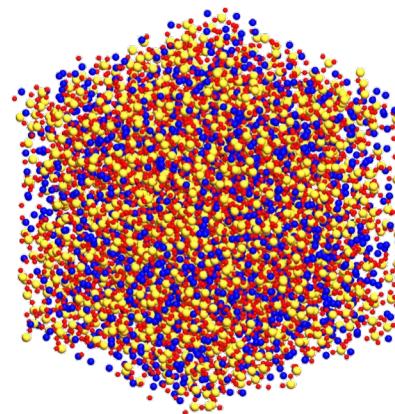
## Silicate glasses

(Zachariasen, 1932)



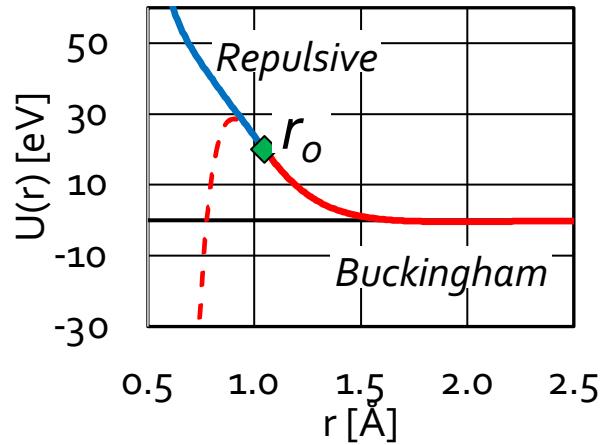
Atomic-resolution electron spectroscopy (Huang et al., 2012)

## Atomic scale simulations



1. BKS Interaction potential
2. Initial sample generation  
(random → heat → quench →  $\text{SiO}_2\text{-xNa}_2\text{O}$ )
3. Verification  
(diffraction, NMR, Brillouin)
4. Athermal deformation

## Potential function



$$U_{BKS}(r_{ij}) = k \frac{q_i q_j}{r_{ij}} + e^{-r_{ij}/\rho} - \frac{C}{r^6}$$

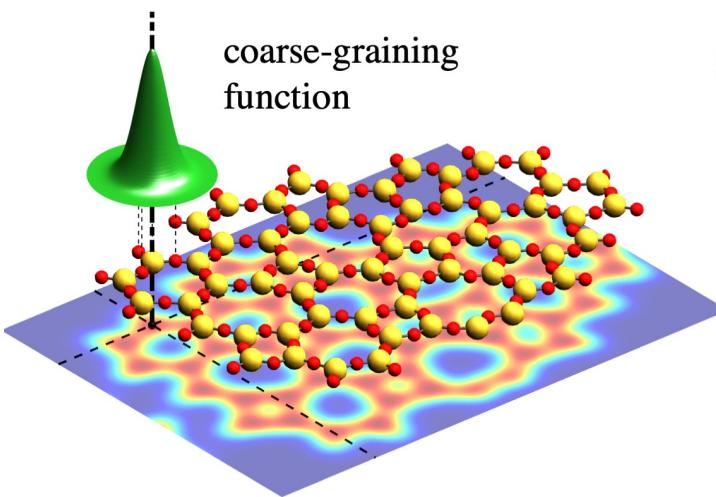
### Coulomb:

- PPPM
- Wolf truncation

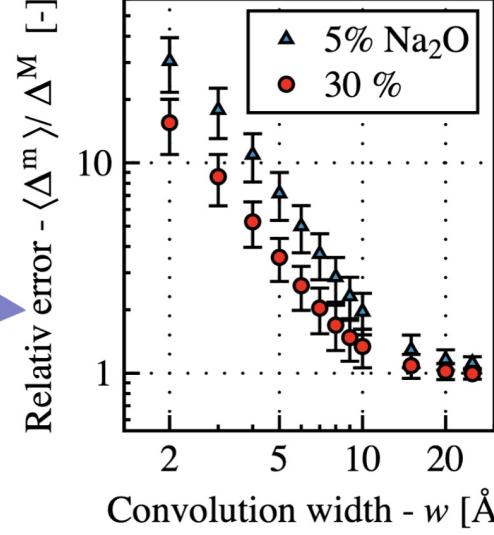
# Resistance of Silicate Glasses

## From discrete to continuum / CG

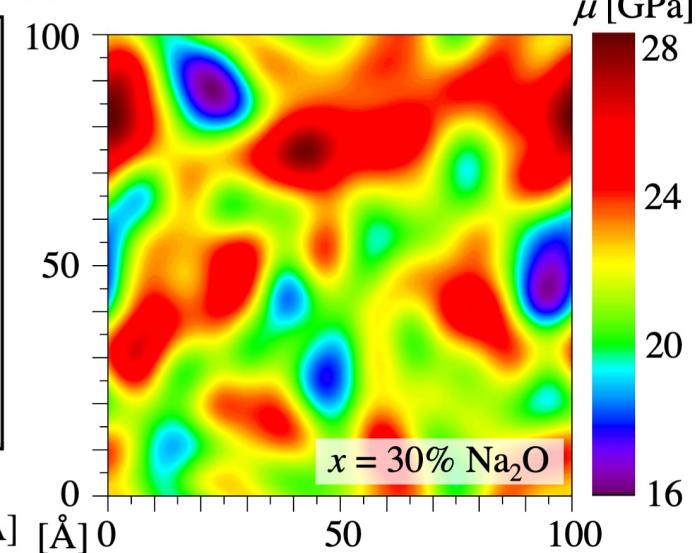
(a) Gaussian convolution



(b) Local error



(c) Local shear modulus

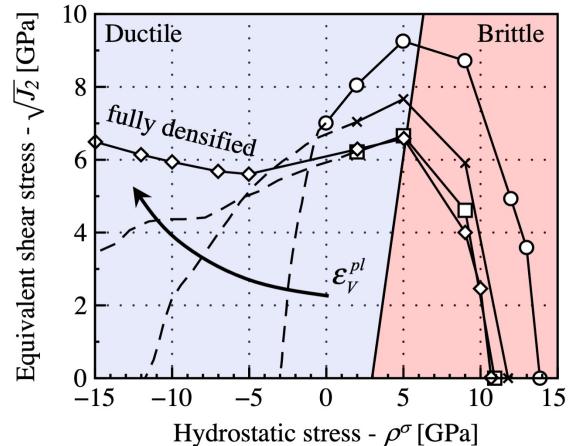
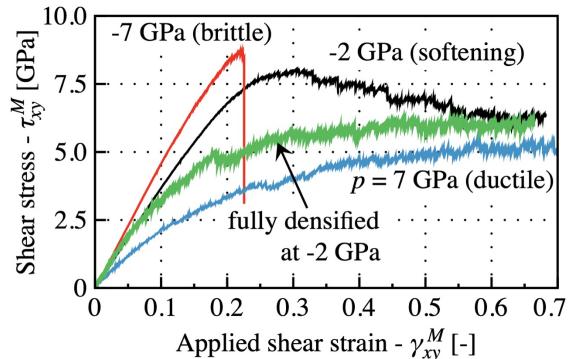


# Resistance of Silicate Glasses

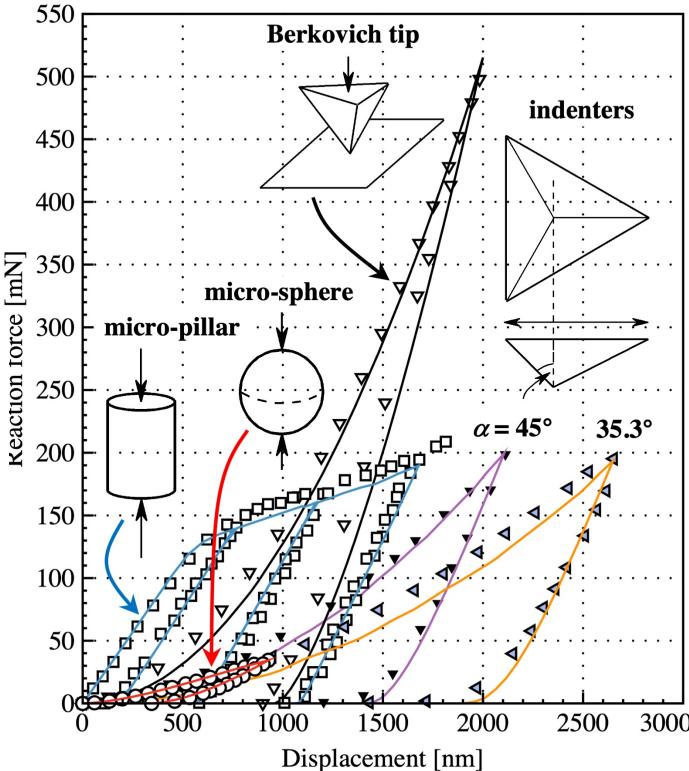
(Kermouche et al., 2016)

## Ductile behavior / Yield criteria

Yield strength: 5-7 GPa



### Experimental comparison



### Micro-pillar

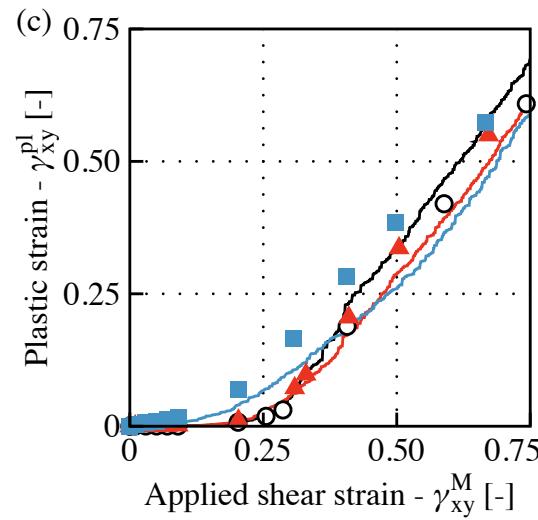
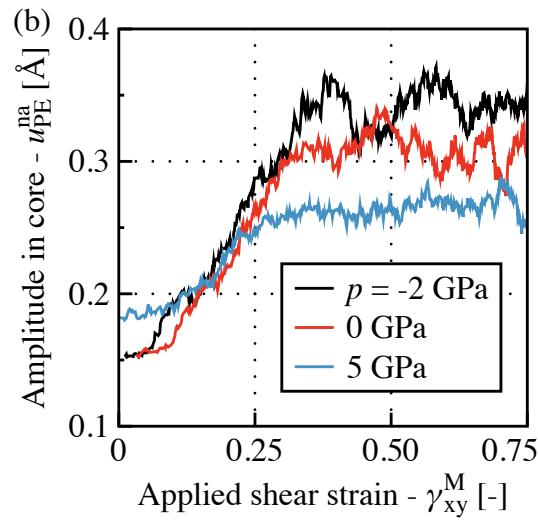
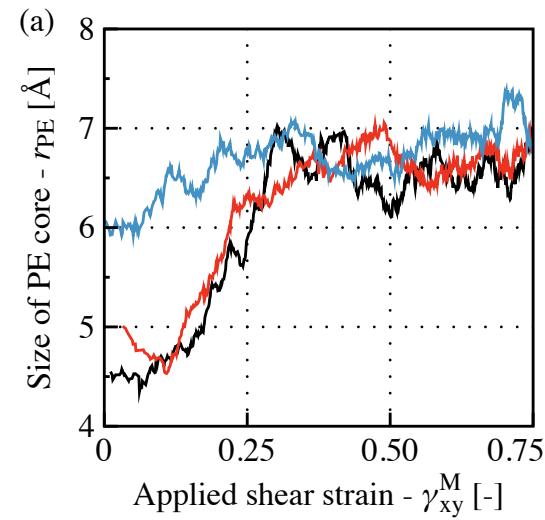
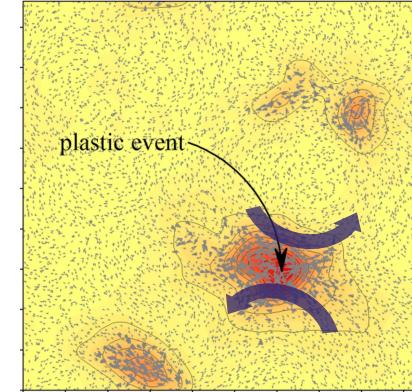
- pure  $\text{SiO}_2$
- ambient temperature
- 4-6  $\mu\text{m}$  size

# Resistance of Silicate Glasses

## Ductile behavior / plastic events

### Plastic event (STZ)

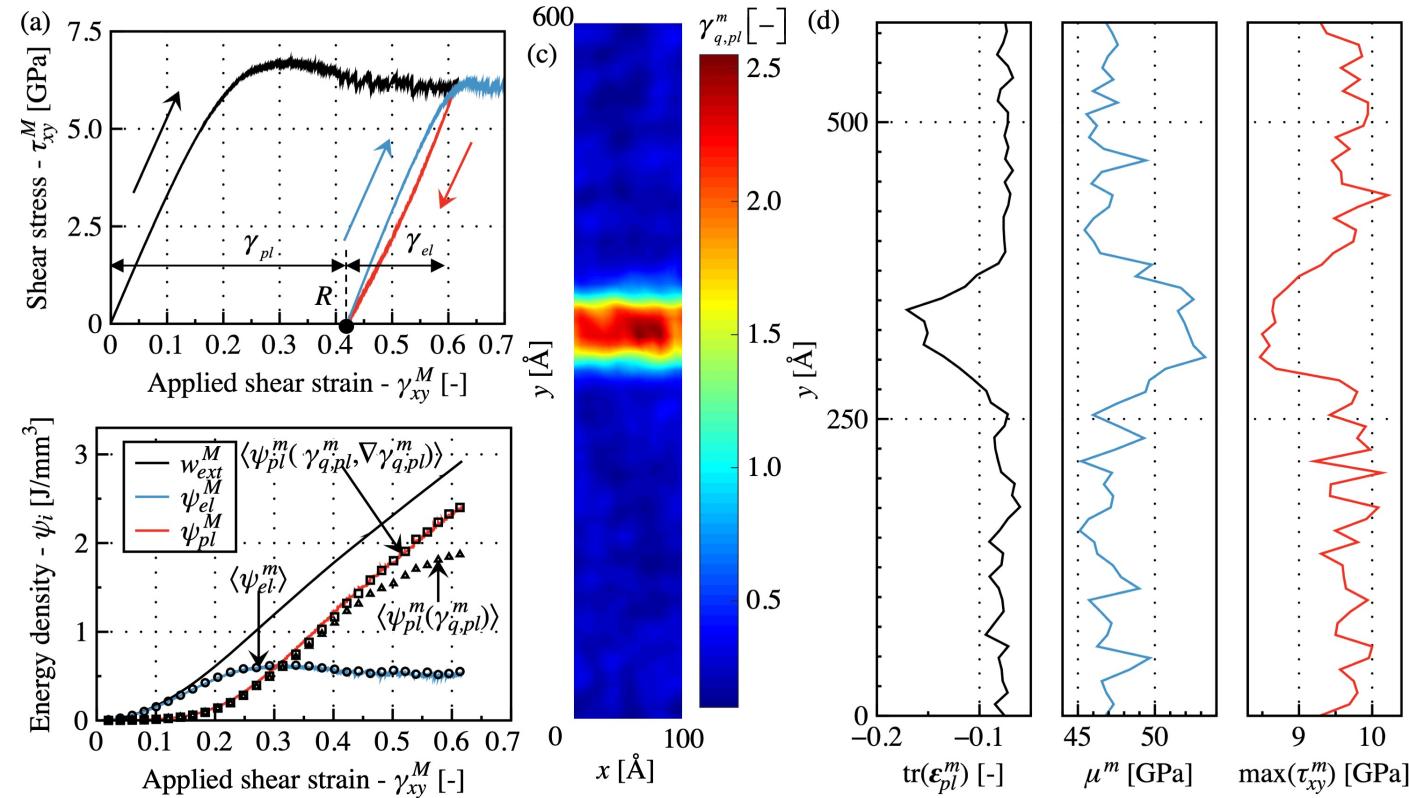
- elliptic shape
- exponential decay
- characteristic size



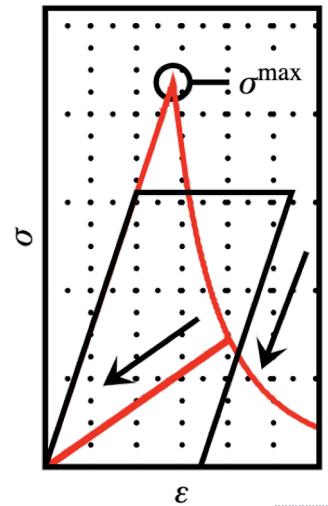
# Resistance of Silicate Glasses

## Ductile behavior / shear banding

$$\langle \psi_{pl}^m \rangle = \frac{1}{V} \int_{\Omega} \left[ \int_t \sigma^m : \varepsilon_{pl}^m dt + l_p^2 \sigma^m |\nabla p_{eq}|^2 \right] d\Omega$$



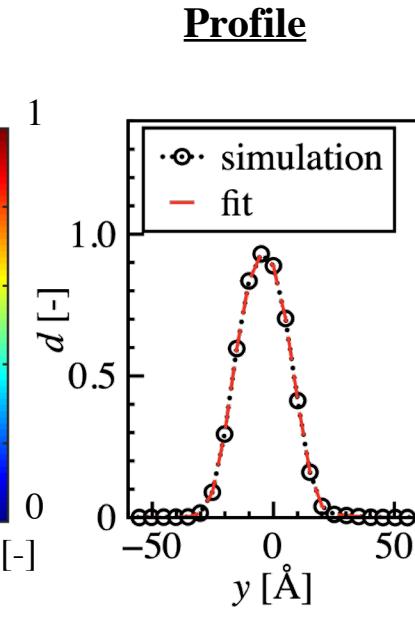
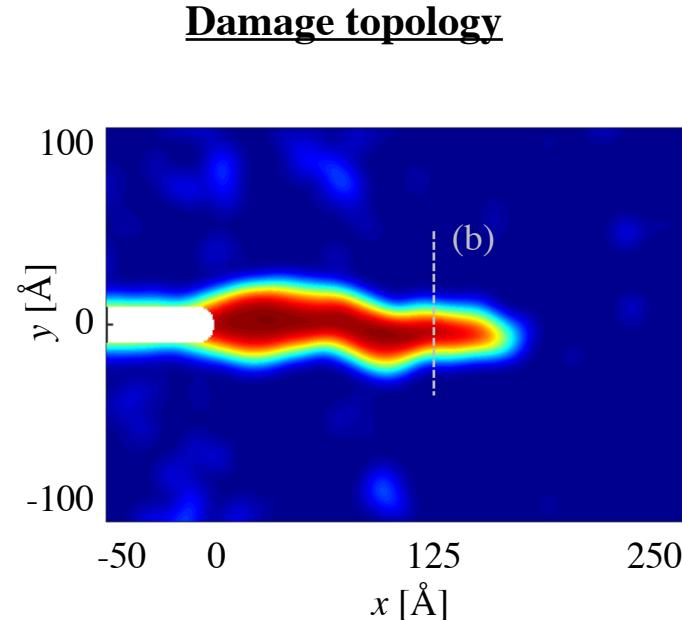
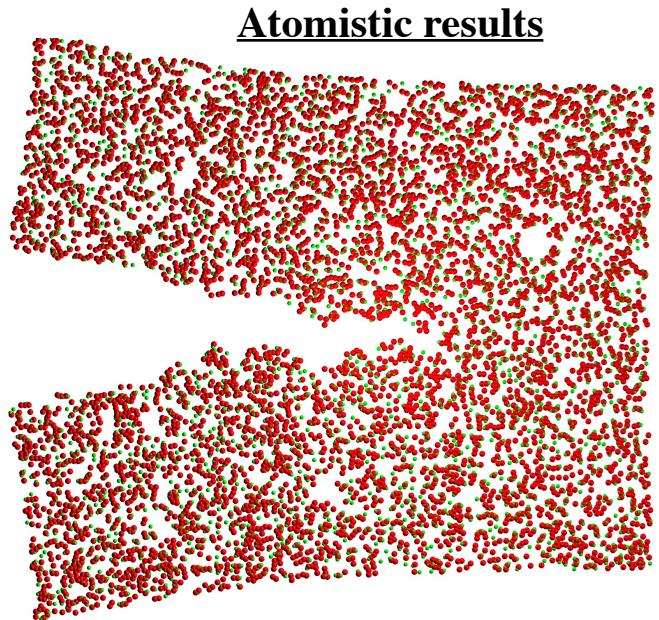
— damaging  
— plastic



# Resistance of Silicate Glasses

(Molnár & Barthel,  
submitted to PRL)

## Fracture

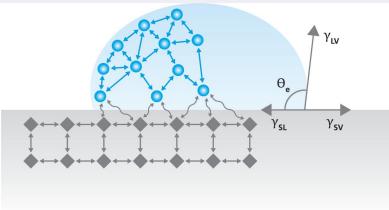


$$\Pi(\mathbf{u}, \mathbf{d}) = \int_{\Omega} (1 - \mathbf{d})^2 \psi_0^+(\boldsymbol{\varepsilon}, \mathbf{C}_0) + \psi_0^-(\boldsymbol{\varepsilon}, \mathbf{C}_0) d\Omega + \frac{g_c}{c_\omega} \int_{\Omega} \left( \frac{\omega(\mathbf{d})}{l_c} + \frac{l_c}{2} |\nabla \mathbf{d}|^2 \right) d\Omega$$

# Resistance of Silicate Glasses

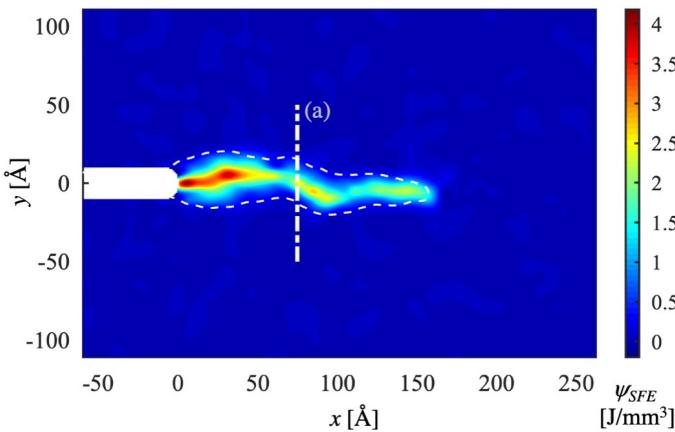
(Molnár & Barthel,  
submitted to PRL)

## Fracture



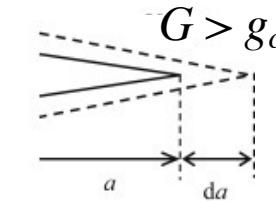
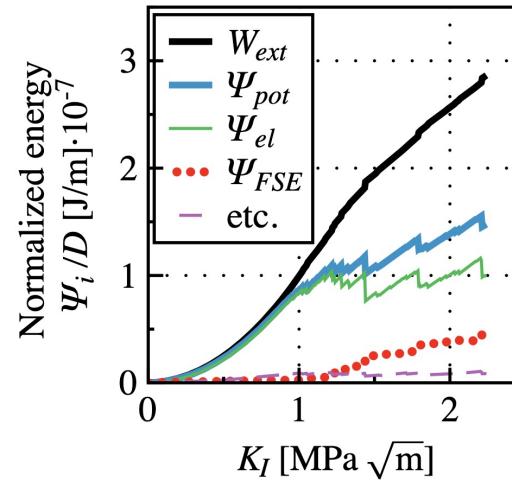
Free Surface Energy    <<    Fracture Toughness

### Free surface energy

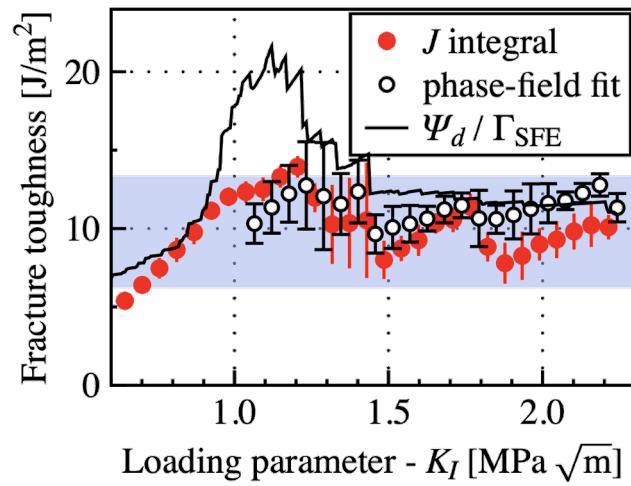


$$\Psi_{FSE} = \Psi_{pot} - \Psi_{el}(\varepsilon)$$

### Energy equilibrium



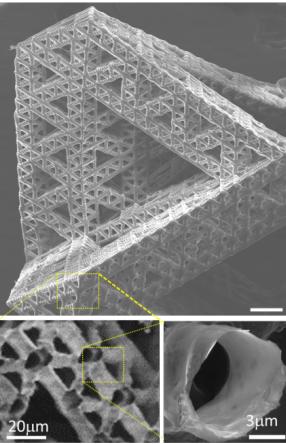
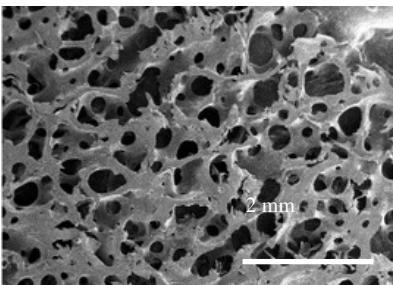
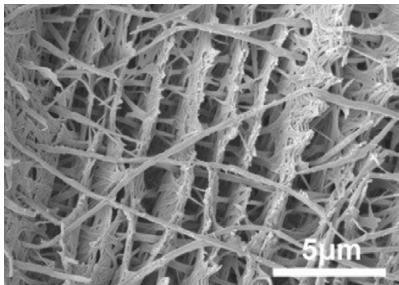
### Fracture toughness



# Architected Materials

## Mechanical metamaterials

### Natural examples

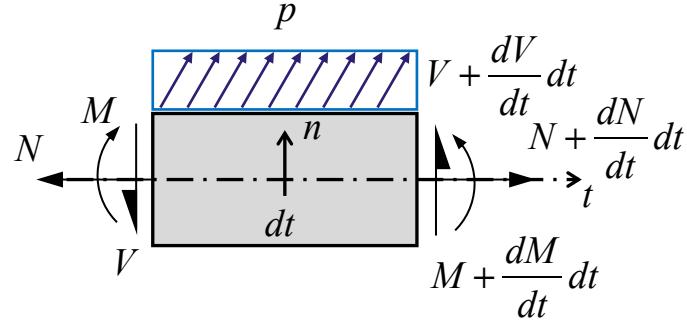


(Lin et al., 2021)  
(Kytyr et al., 2012)

### Man-made examples

(Zheng et al, 2015)

### Euler-Bernoulli beam theory



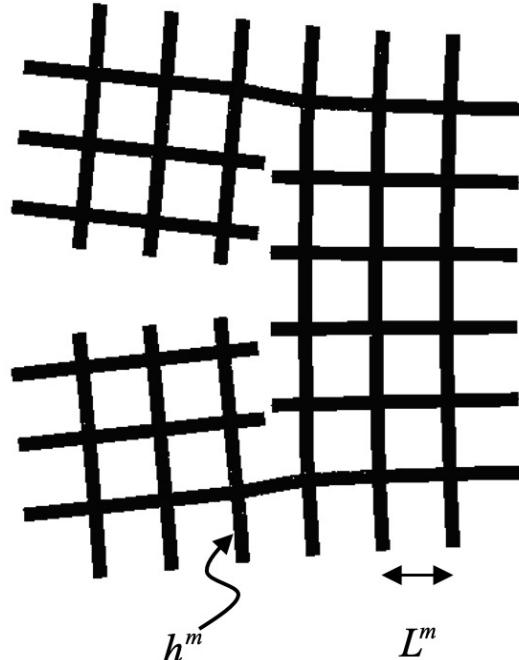
### Modelling assumptions

- beams are flawless
- no stress concentration at joints
- $\sigma_t < \sigma_c^m$

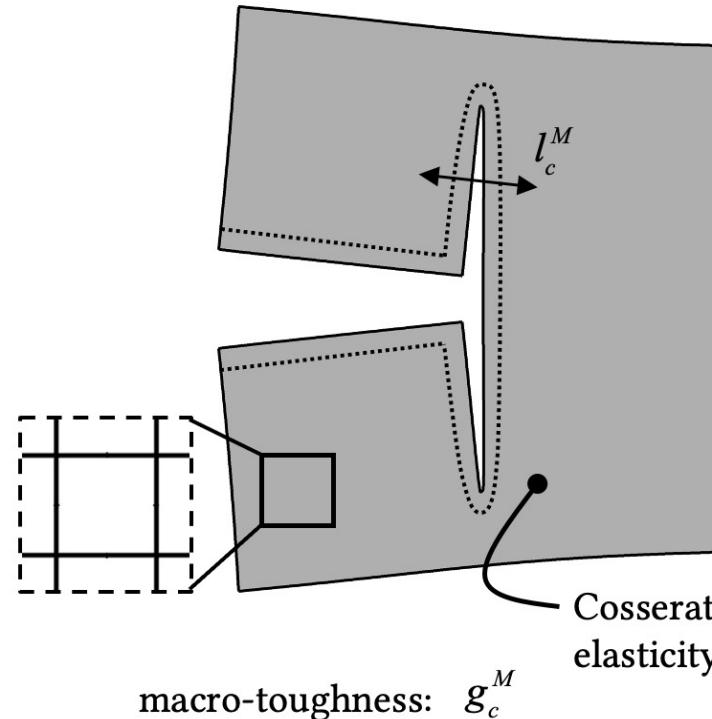


## Strategy

(a) Discrete beam



(b) Continuum phase-field



- Critical load
- Fracture topology
- Various loading

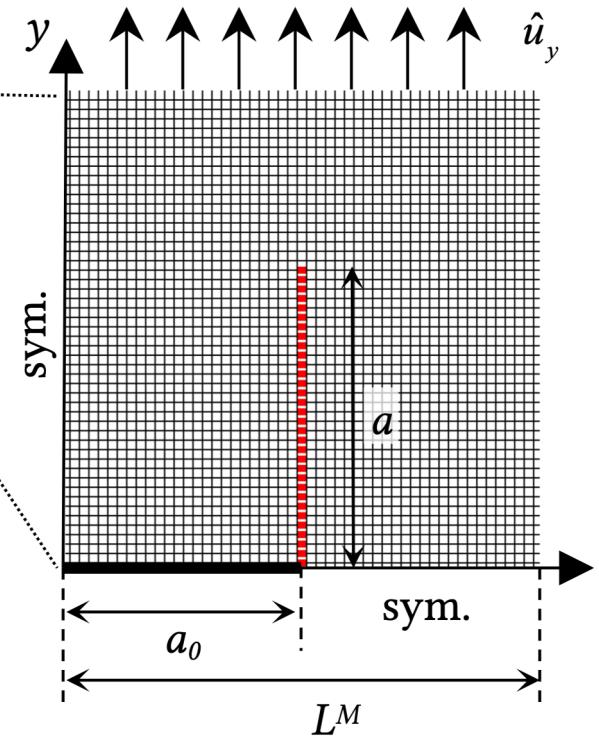
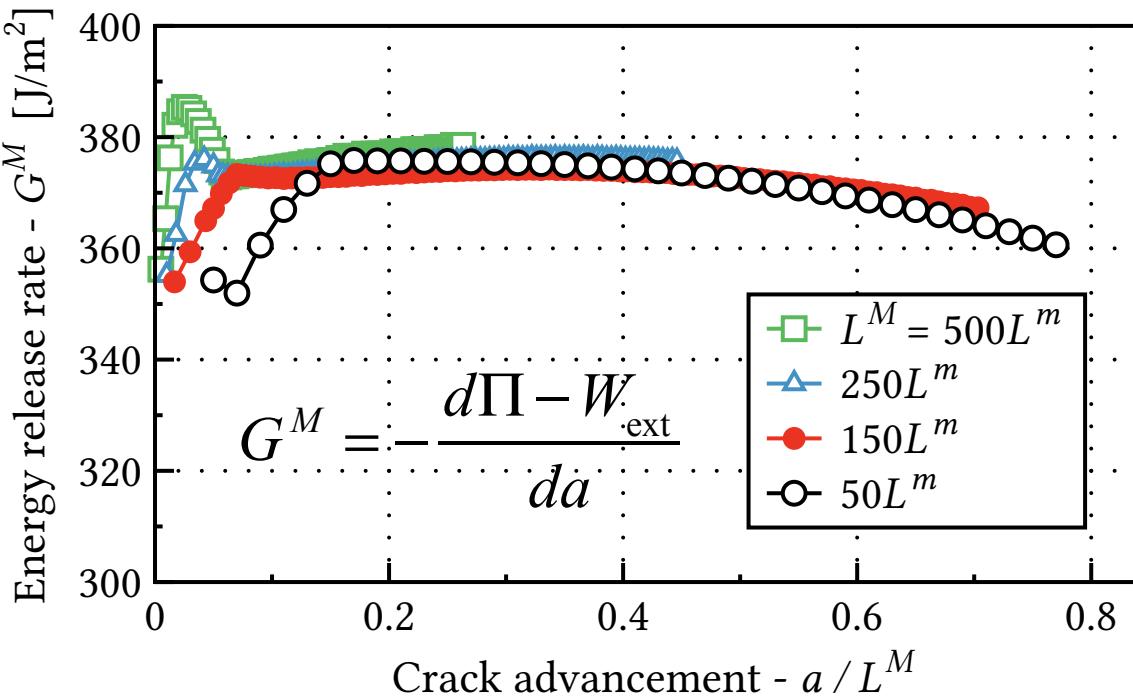
micro-strength:  $\sigma_c^m$

macro-toughness:  $g_c^M$

# Architected Materials

(Molnár & Réthoré,  
submitted to JTCAM)

## Existence of a unique toughness



# Continuum model

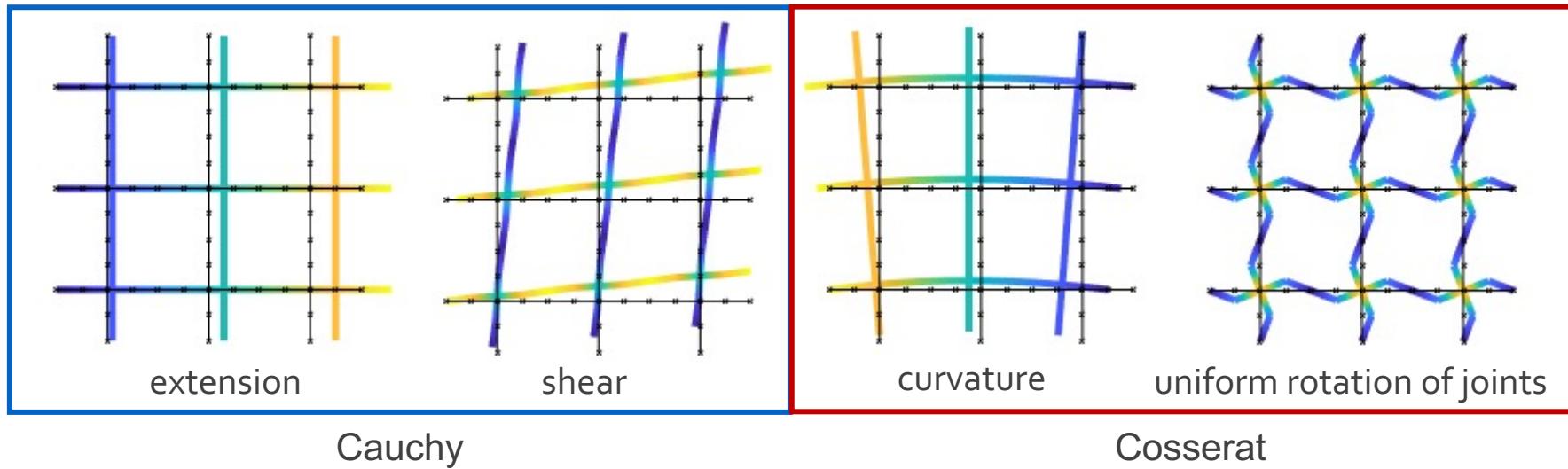
(Molnár & Réthoré,  
submitted to JTCAM)

## Cosserat phase-field fracture

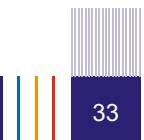
New DOF: rotation  $\phi$

New deformations

(Cosserat & Cosserat, 1909)

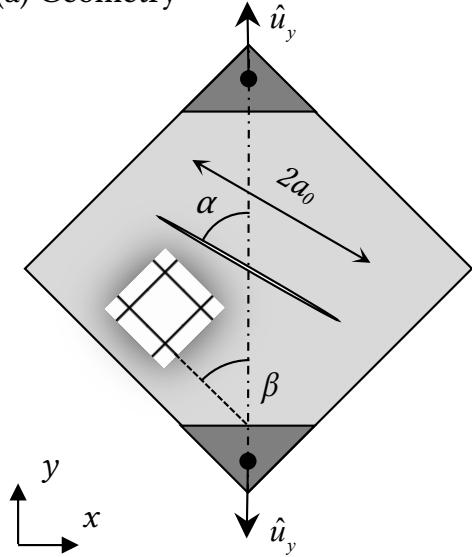


$$\Pi(\mathbf{u}, \mathbf{d}) = \Psi(\mathbf{u}, \phi, g_{ij}(\mathbf{d}_i)) + \sum_{i=1}^2 \frac{3g_{c,M,i}}{8l_{c,M,i}} \int_{\Omega} \left( \mathbf{d}_i + l_{c,M,i}^2 |\nabla \mathbf{d}_i|^2 \right) d\Omega$$

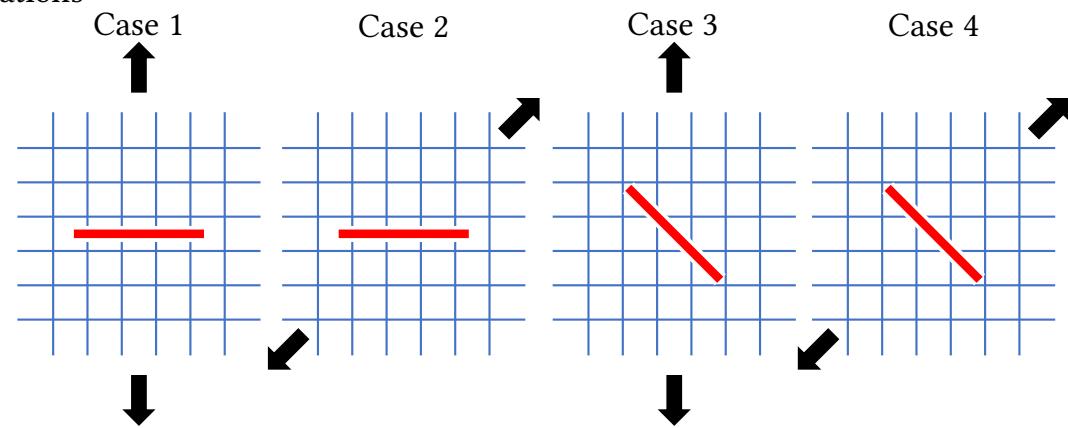


## Experimental validation

(a) Geometry



(b) Orientations



Crack/Loading ( $\alpha$ ):	90°
Structure/Loading ( $\beta$ ):	90°
Crack/Structure :	0°

Crack/Loading ( $\alpha$ ):	45°
Structure/Loading ( $\beta$ ):	45°
Crack/Structure :	0°

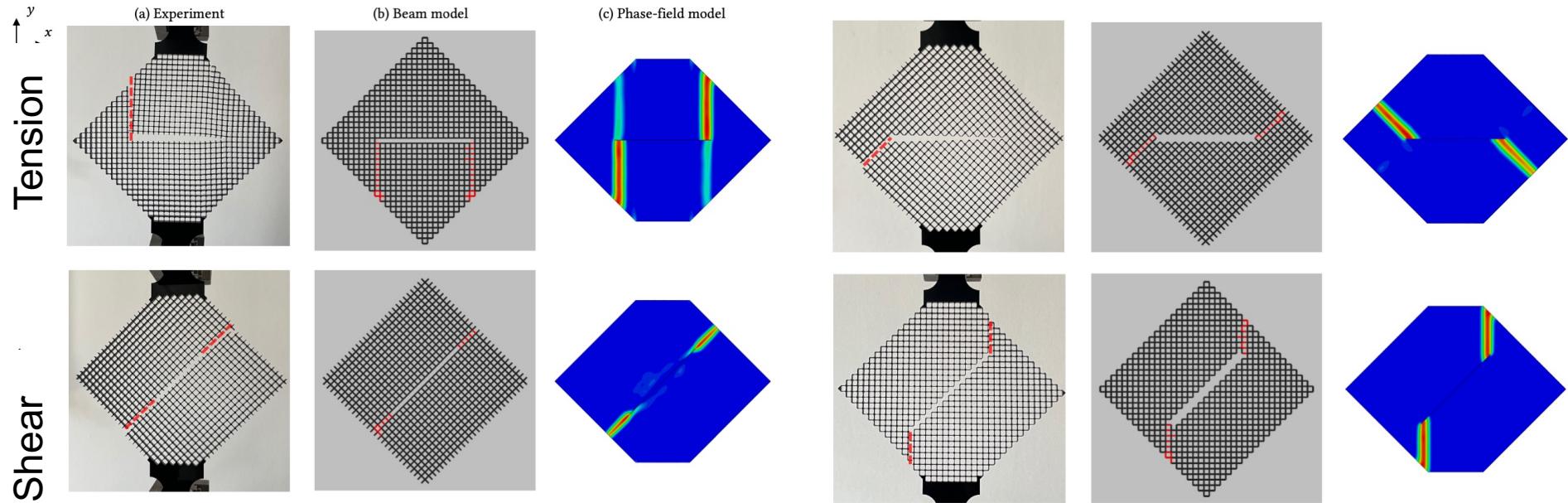
Crack/Loading ( $\alpha$ ):	45°
Structure/Loading ( $\beta$ ):	90°
Crack/Structure :	45°

Crack/Loading ( $\alpha$ ):	90°
Structure/Loading ( $\beta$ ):	45°
Crack/Structure :	45°

# Architected Materials

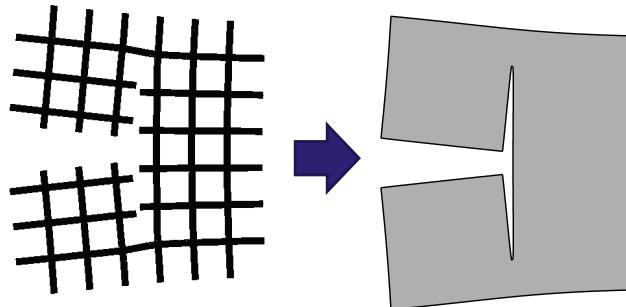
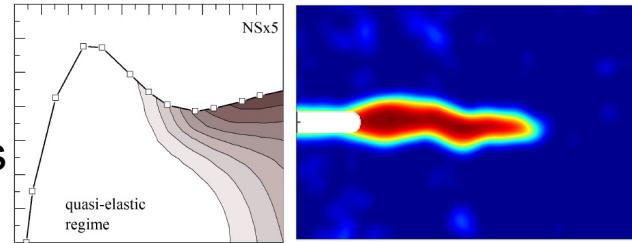
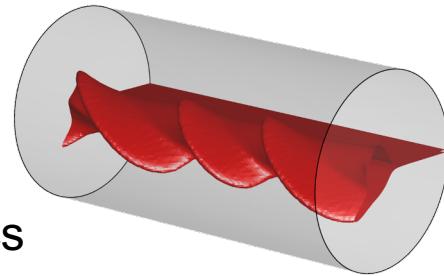
(Molnár & Réthoré,  
submitted to JTCAM)

## Experimental validation



# Contributions

1. **Regularization** allows us to model **real** physical phenomena
2. Irwin's length – phase-field length – characteristic crack length
3. Connection of **limiting velocity** and **regularization** in dynamics
  
4. **Length scales** form the **atomic scale** for glasses
5. **Damage** correlates **free** and **fracture** surface energies
  
6. Periodic **beam lattices** have a **unique fracture toughness**
7. **Cosserat** theory is necessary and sufficient.



# Contributions

## Open-source phase-field implementation

The screenshot shows a website header with logos for CNRS, G Research, and INSA Lyon. Below the header is a navigation bar with links: HOME, ABOUT ME, RESEARCH ▾, TUTORIALS ▾, PUBLICATIONS, JOB OFFERS, CONTACT, and LINKS. A blue banner below the navigation bar reads "Fracture modeling with phase field method". The main content area contains two sections: "1. Simple tension with 2 elements" and "2. Single edge notched sample & MATLAB converter". Each section includes a brief description and a link for download.

**1. Simple tension with 2 elements**  
The tutorial presents a simple conversion between the input file generated by ABAQUS and the use of the new UEL. The instructions can be downloaded from [here](#). While the files used and created through the tutorial are accessible from [this link](#).

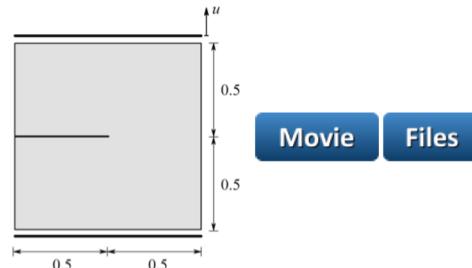
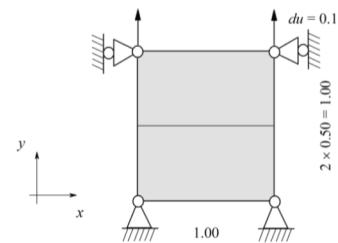
**2. Single edge notched sample & MATLAB converter**  
The tutorial presents a single edge notched sample and how to develop simple models using ABAQUS/CAE and convert them with a MATLAB script (still in Beta testing stage). It contains a FORTRAN script with both triangular and square elements. The instructions are shown [here](#). While the files used and created through the tutorial are accessible from [this link](#).



Please visit

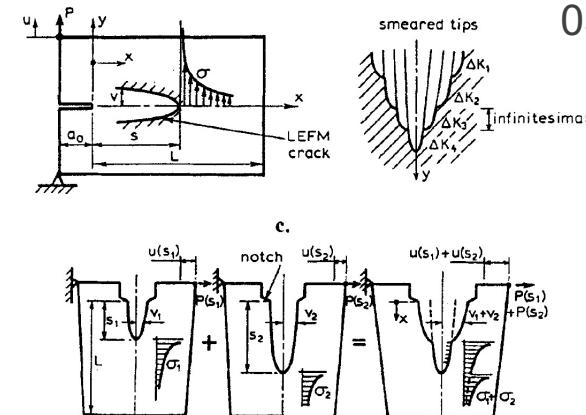
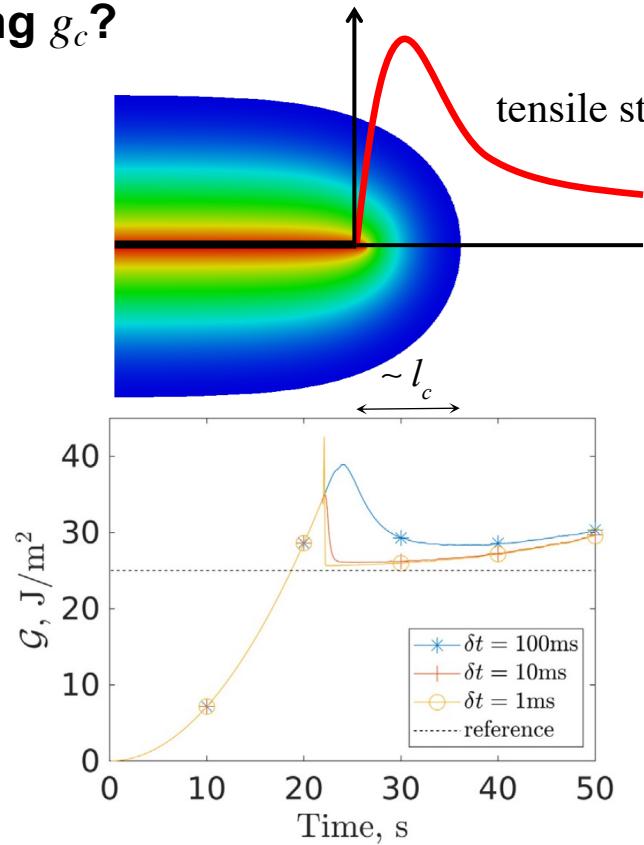
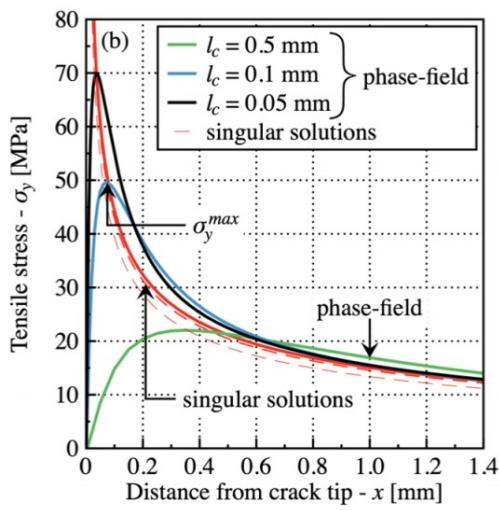
[www.molnar-research.com](http://www.molnar-research.com)

gergely.molnar@insa-lyon.fr



# Perspectives

## Measurement of $l_c$ ang $g_c$ ?



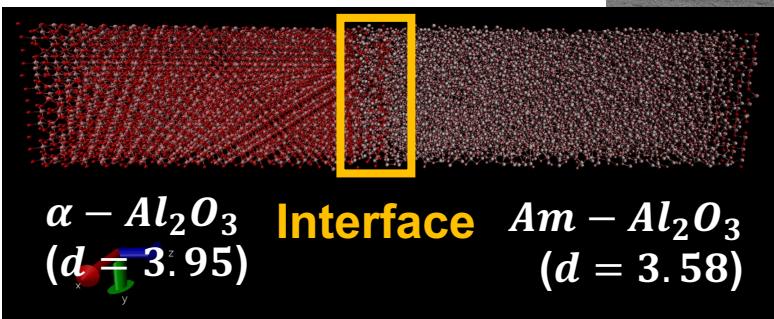
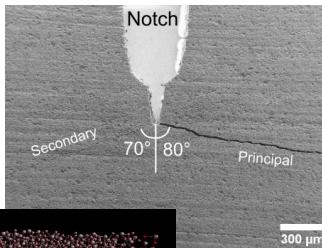
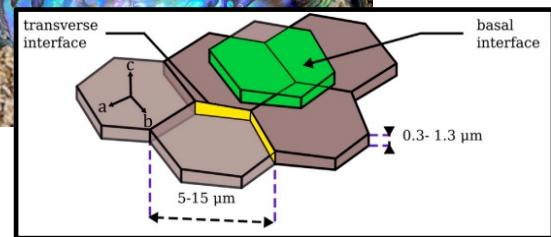
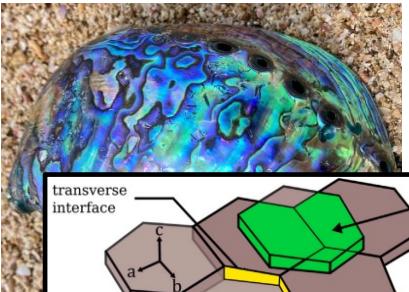
# Perspectives

## Atomic-scale modeling

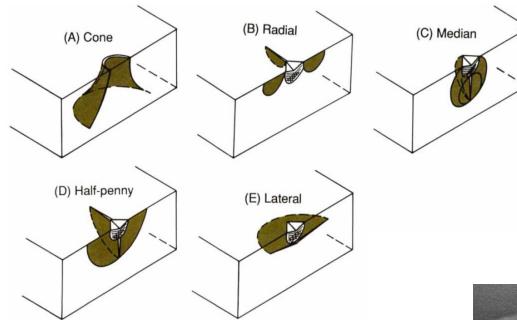
with A. Doitrand  
S. Meille



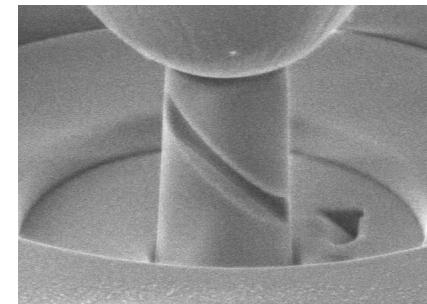
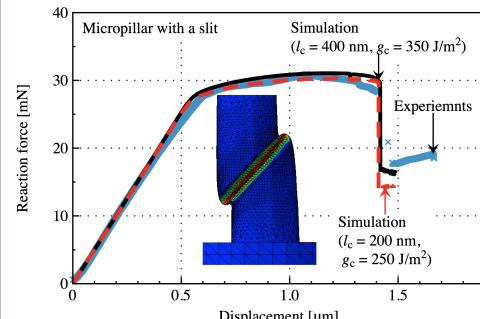
Benjamin Bert  
10/2023



## Indentation resistance of silicate glasses



(Cook & Pharr, 1990)



(Guillonneau et al.,  
2022)

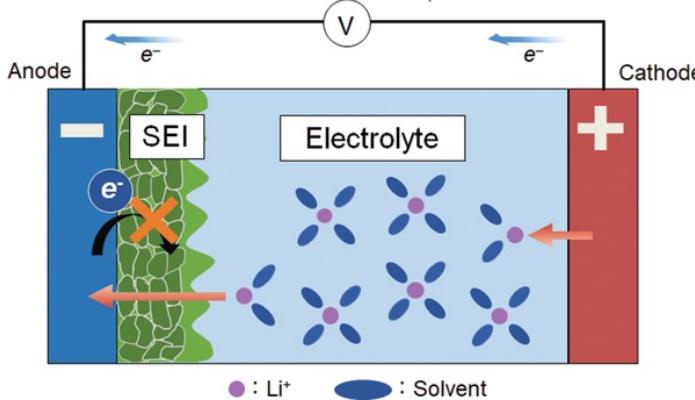


Souhail Chaouch  
05/2025  
anr  
GaLAaD

# Perspectives

## Si based battery optimization

(Tanaka et al., 2021)



### Advantages

- 10x capacity
- natural, abundant, cheap
- environmentally friendly

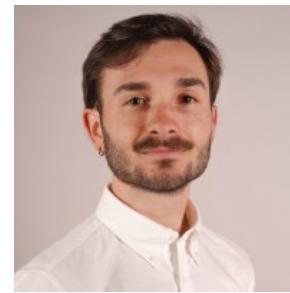
### Disadvantages

- 300% volume expansion
- few charges (delamination)

with N. Blal  
D. Machon

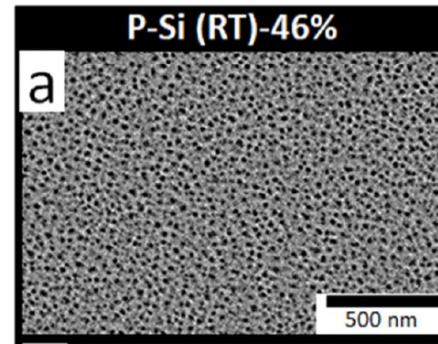
UDS

cnrs



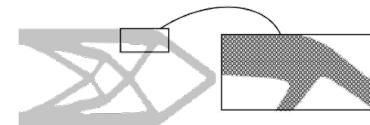
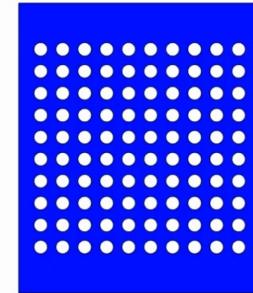
Tom Guisard  
10/2024

### Post-treatment



(Abdelouhab et al., 2024)

### Optimization



G. Molnár

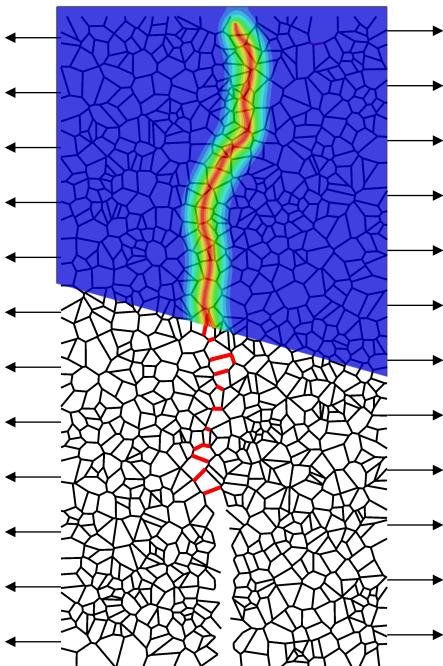


40

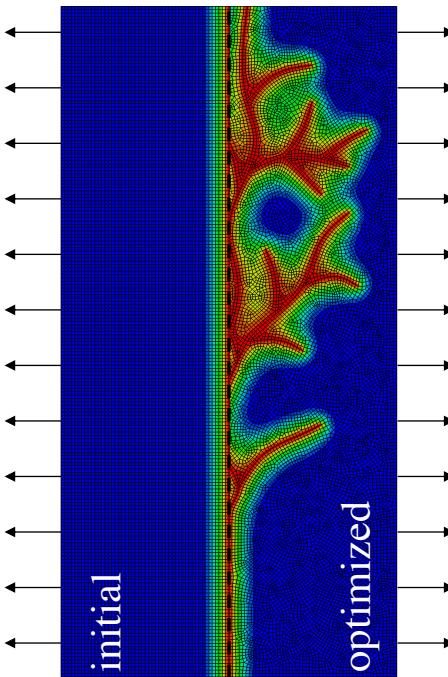
# Perspectives

## Optimization for toughness

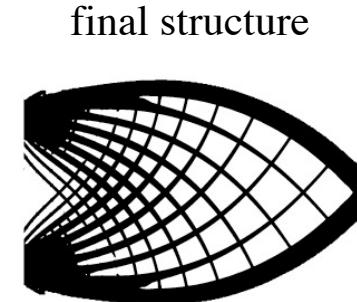
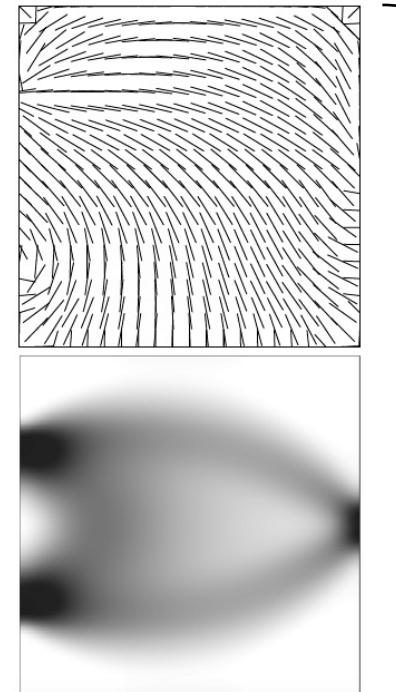
(a) Homogenization



(b) Optimization



(c) Rendering and testing



# Acknowledgement

## Mentors



## PhD and postdocs



## Jury

Samuel Forest, David Rodney,  
Jean-François Molinari, Nicolas Moës, Laura De Lorenzis

and S. Fournier, M. Sepulveda,  
K. Daigne, R. Alvarez-Donado  
B. Bert, T. Guisard

## Collages

R. Estevez, D. Leguillon, N. Blal, N. Tardif, B. Prabel,  
M.-C. Baietto, M. Coret, D. Machon, R. Seghir,  
C. Martinet, T. Deschamps, S. Meille,  
G. Kermouche, P. Ganster, L. Orgéas, A. Tanguy,  
V. Lazarus, F. Morestin, S. Patinet



## My Family

# Acknowledgement

## Aurélien Doitrand

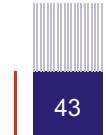


Pot: 2ème étage  
de la Bibliothèque

Save the date!

28/03/2025

G. Molnár



Laboratoire MATéIS : Matériaux, Ingénierie et Science - UMR 5510